



## Accelerating multiscale simulations with surrogate models based on recurrent neural networks

Moisés Zarzoso Carlos González









## Simulation of Composite Materials Coupling Scales RNN Learning Plastic Behaviour Learning Damaged Behaviour Summary and Future Work



## **Simulation of Composite Materials**

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Macroscale

Microscale  $\sim 10^{-6} m$ 





LLorca, J., González, C., Molina-Aldareguía, J.M., Segurado, J., Seltzer, R., Sket, F., Rodríguez, M., Sádaba, S., Muñoz, R. and Canal, L.P. (2011), Multiscale Modeling of Composite Materials: a Roadmap Towards Virtual Testing. Adv. Mater., 23: 5130-5147.

Mesoscale



## **Simulation of Composite Materials**

Microscale  $\sim 10^{-6} m$ 

Matrix Cracking Fibre Breakage Fibre Kinking Interface Decohesion

Mesoscale  $\sim 10^{-2} m$ 

Delamination

Macroscale Failure  $\sim 10^1 - 10^2 m$ 



LLorca, J., González, C., Molina-Aldareguía, J.M., Segurado, J., Seltzer, R., Sket, F., Rodríguez, M., Sádaba, S., Muñoz, R. and Canal, L.P. (2011), Multiscale Modeling of Composite Materials: a Roadmap Towards Virtual Testing. Adv. Mater., 23: 5130-5147.



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#### MULTISCALE SIMULATION STRATEGY





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### **Simulation of Composite Materials**



A. H. Baluch, O. Falcó, J. L. Jiménez, B. H.A.H. Tijs, C. S. Lopes,

This project has received funding from the European Union's Horizon 2020 An efficient numerical approach to the prediction of laminate tolerance to Barely Visible Impact Damage, 6 composite Structures, Volume 225, 2019



## FE<sup>2</sup>- Concurrent



**Computationally Expensive** 

Representative Volume Elements are simple





**Coupling Scales.** Surrogate model based on NNs





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#### Elastic

 $\boldsymbol{\sigma}(t_i) = f\left(\boldsymbol{\varepsilon}(t_i)\right)$ 

**Damaged Behaviour** 

 $\boldsymbol{\sigma}(t_i) = f\left(\boldsymbol{\varepsilon}(t_0, t_1, \dots, t_{i-1}, t_i)\right)$ 



### **Recurrent Neural Networks**

Elastic

 $\boldsymbol{\sigma}(t_i) = f\left(\boldsymbol{\varepsilon}(t_i)\right)$ 

**Damaged Behaviour** 

 $\boldsymbol{\sigma}(t_i) = f\left(\boldsymbol{\varepsilon}(t_0, t_1, \dots, t_{i-1}, t_i)\right)$ 



$$W_{xh} \in R^{h_{units} \times n_x}$$
$$W_{hh} \in R^{h_{units} \times h_{units}}$$
$$W_{yh} \in R^{n_y \times h_{units}}$$

$$\mathbf{h}_{t} = tanh \left( W_{xh} \mathbf{x}_{t} + W_{hh} \mathbf{h}_{t-1} \right)$$
$$\mathbf{y}_{t} = W_{hy} \mathbf{h}_{t}$$

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Initialization :  $\mathbf{h}_0 = ar{0}$ 





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#### **Intermediate Modulus Carbon Fibres**

Transversally	$\nu_T$	$E_{T}[GPa]$
Isotropic	0.40	13.00

#### **Epoxy Resin**

E[GPa]	ν	Isotropic
5.07	0.35	



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#### **Dataset Generation.** Generation of strain history paths



Abaqus (Implicit Solver) 0.05  $\varepsilon_{21}$  $t \in [0,1]$ -0.05  $\varepsilon \in [-0.1, 0.1]$ -0.1 0 0.1

0.1

1000 paths







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0.2

0.3

0.4

0.5

0.6

0.7

0.8

0.9

### Dataset Generation. Accumulated Plastic Strain





- +5.150e+00
- +4.721e+00 - +4.292e+00
- +3.863e+00
- +3.004e+00
- +2.575e+00 - +2.146e+00
- +1.717e+00
- +8.583e-01
- +4.292e-01 - +0.000e+00

t = 0









t = 1

t = 0.75



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#### Dataset Generation. Effect of Fibre's Distribution

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### Neural Network. Architecture







### **Results.** Model Validation





Pure Elastic Behaviour







Matrix-Fibre Interface: **Cohesive Elements** 

Matrix:

 $\varepsilon_{X,1}$ 

Plasticity Damage

#### **Periodic Boundary Conditions**

Strain Paths: *t* ∈ [0,1]  $\varepsilon \in [-0.012, 0.012]$ 



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## **Damage Evolution**

#### 

#### Tensile Damage



t = 0.16



t = 0





t = 0.05



t = 0.83



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Surro

### **Learned Behaviour**





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- A methodology for training surrogate models for RVEs is tested on a micro-scale composite material RVE with matrix damage and interface debonding
- The proposed method is able to reconstruct general-shape strain-stress curves under damaged conditions.

#### **Future Work**

Implementation in Abaqus

3D RVE

Geometrical Parameters(Volume Fraction, Fibre Diameter,...)





# Thank you for your attention

moises.zarzoso@imdea.org

materials.imdea.org



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