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Dr. Jérémy CHEVALIER

INFLUENCE OF CRYSTALLIZATION CONDITIONS ON THE NANO-/MICRO-BEHAVIOR OF CARBON FIBER-REINFORCED PEEK COMPOSITE

CONTEXT

Thermoset  Thermoplastic

ECONOMIC 

Easier production of
complex parts at
higher rate

PERFORMANCE 

Higher toughness

ENVIRONMENTAL



CONTEXT

Thermoset  Thermoplastic

ECONOMIC

Easier production of complex parts at higher rate

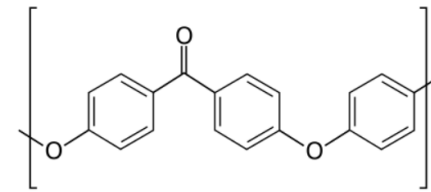
PERFORMANCE

Higher toughness

ENVIRONMENTAL



PEEK/C fiber
(Unidirectional, 66 fiber wt. %)



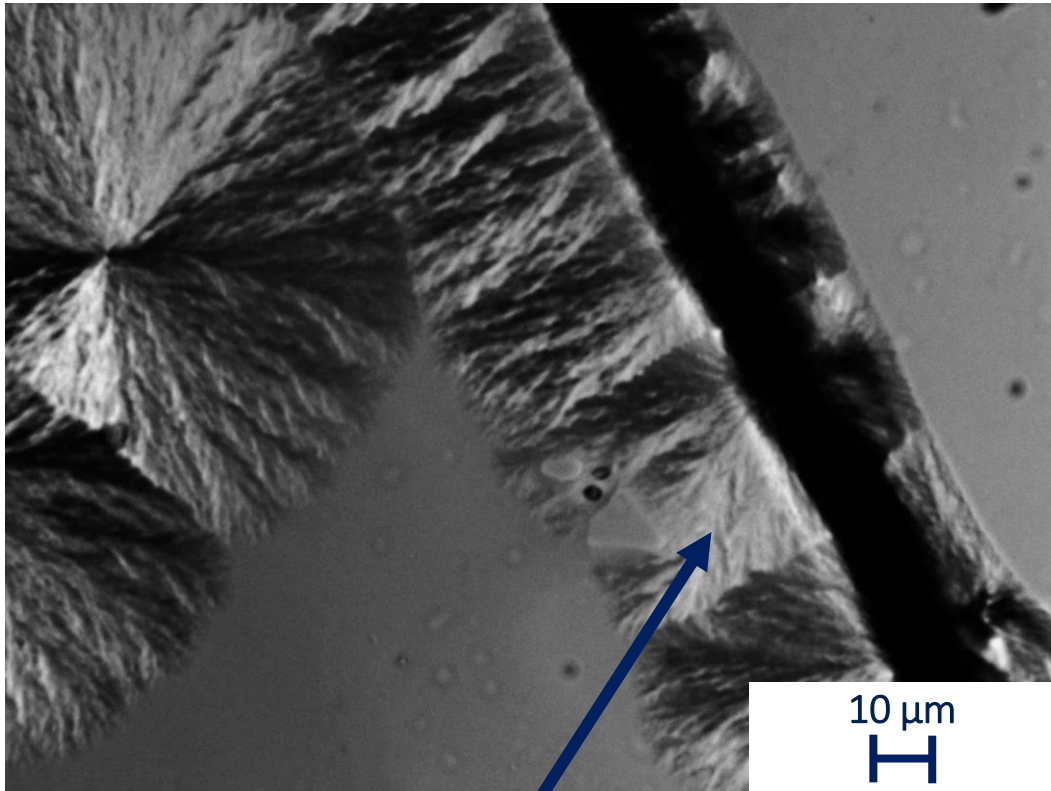
$$T_g \approx 143^{\circ}\text{C}$$
$$T_m \approx 343^{\circ}\text{C}$$

SEMI-CRYSTALLINE THERMOPLASTIC COMPOSITE: LIMITATION & QUESTIONS

Processing conditions



Performances of
thermoplastic composites



Transcrystallization (TC)

In the literature



Influence of processing conditions
(time and temperature) on TC ?



Influence of TC on macro properties
(σ_0 , E and K_{IC})?



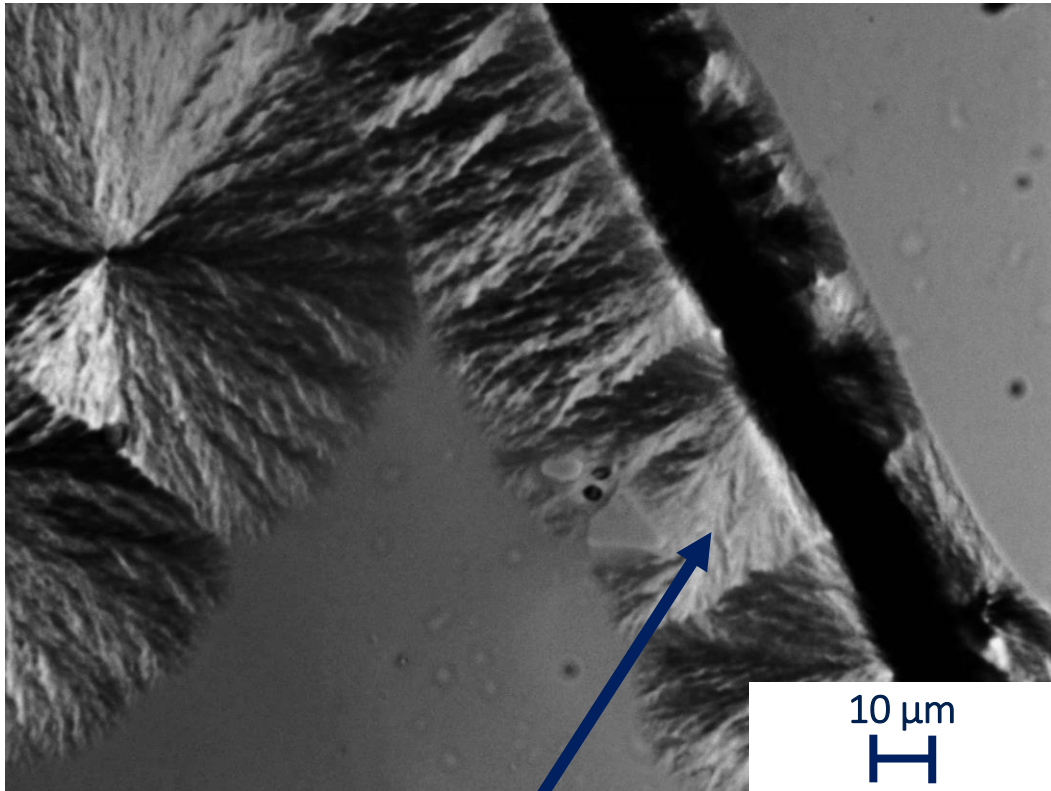
Impact of TC on mechanical behavior in
the interfiber zones?

SEMI-CRYSTALLINE THERMOPLASTIC COMPOSITE: LIMITATION & QUESTIONS

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thermoplastic composites



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Influence of processing conditions
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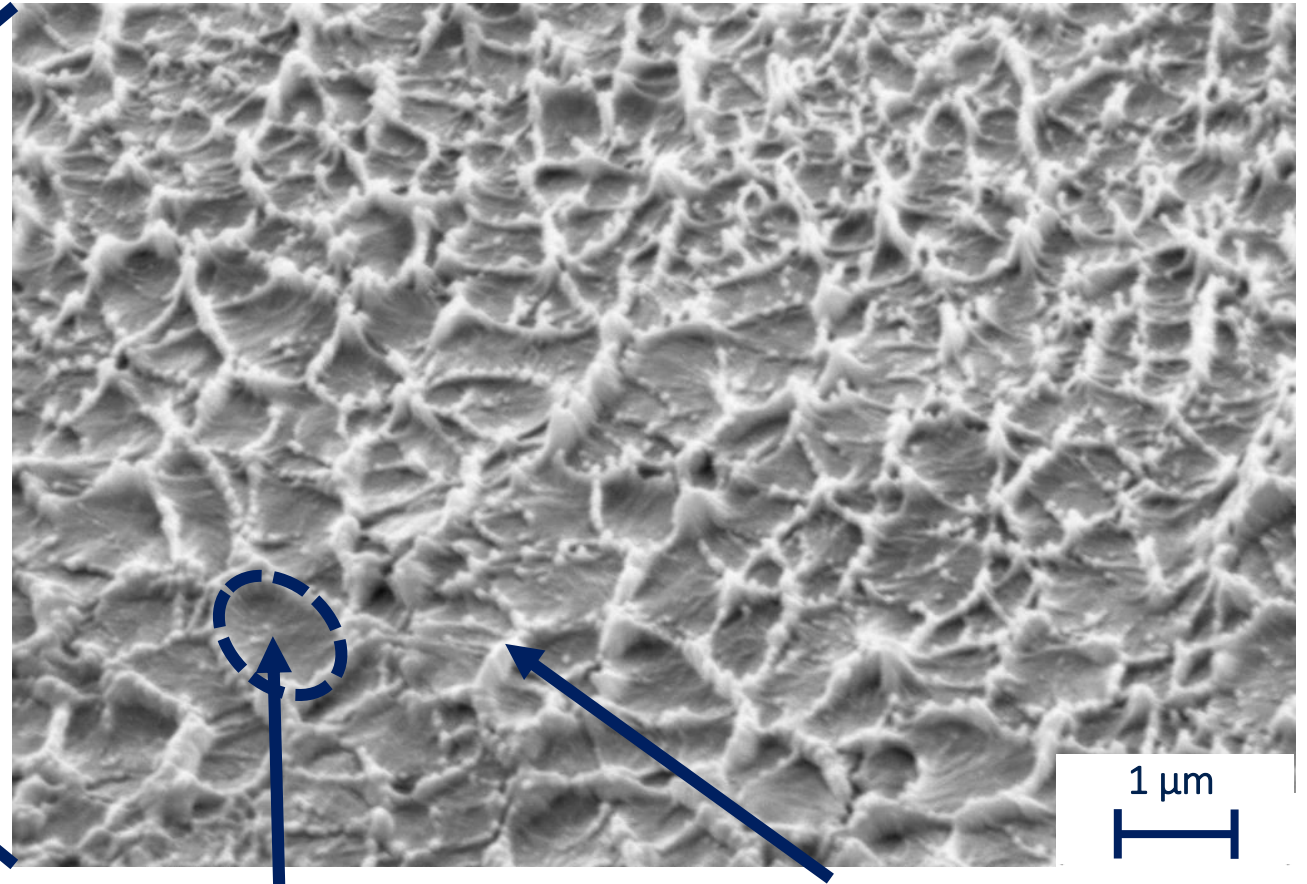
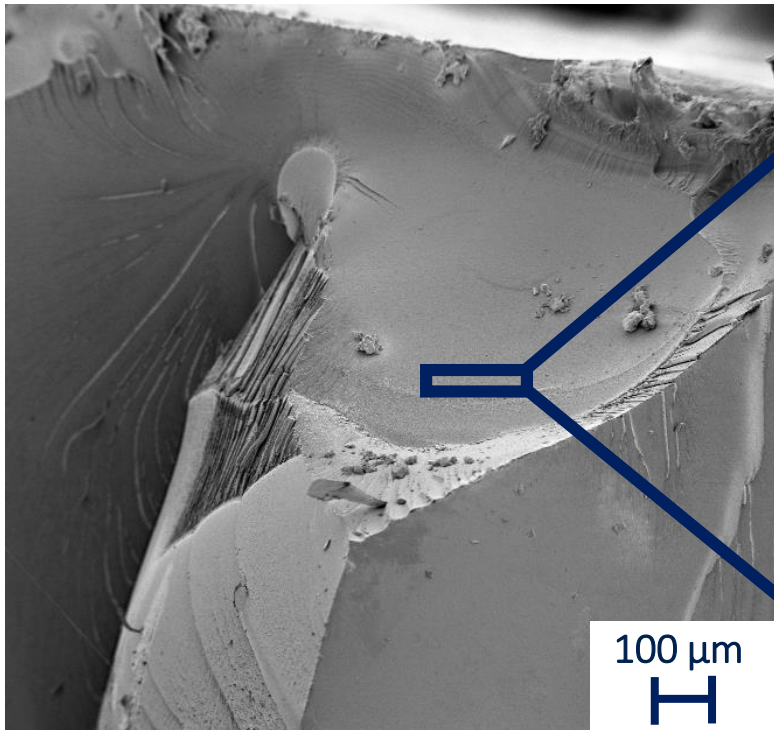
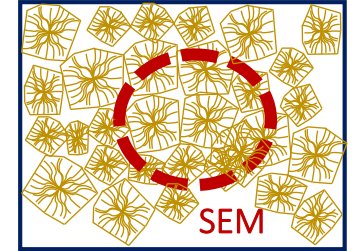
Influence of TC on macro properties
(σ_0 , E and K_{IC})?



Impact of TC on mechanical behavior in
the interfiber zones?

CHARACTERIZATION OF SPHERULITES – Liq. N_2 fracture + SEM

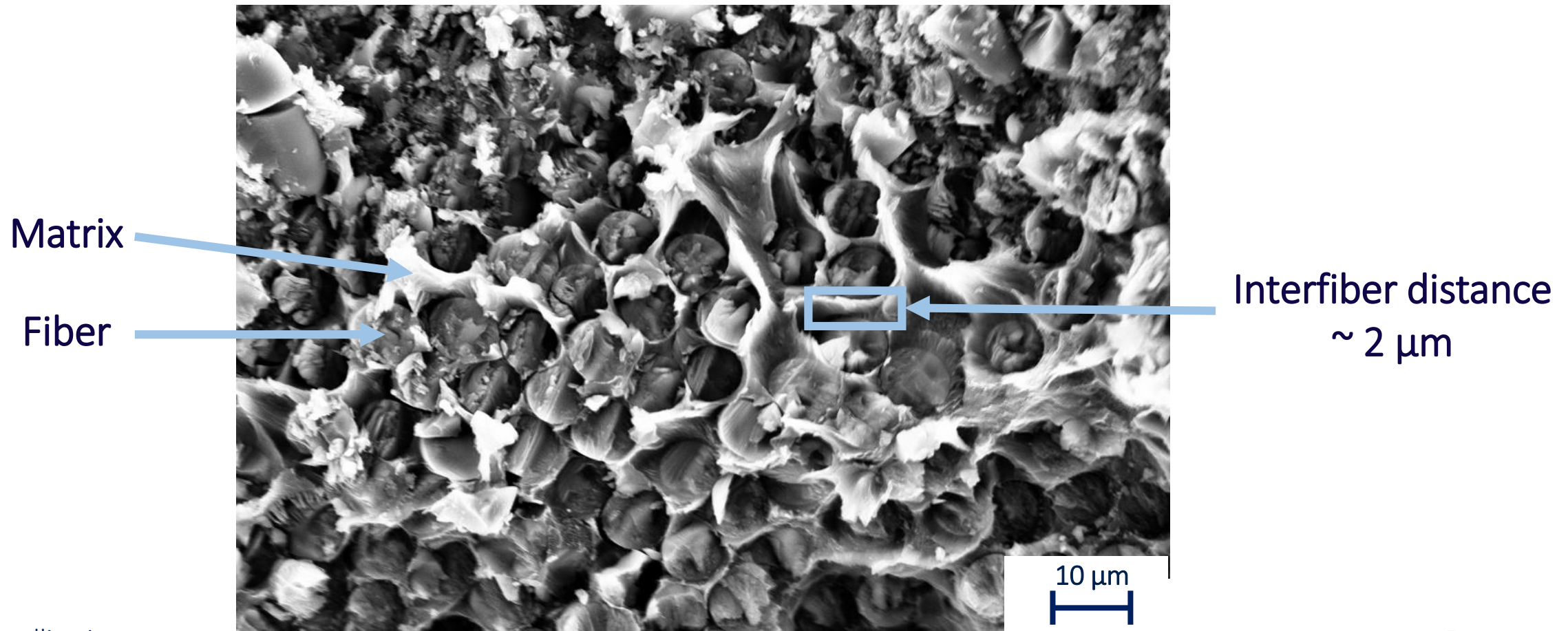
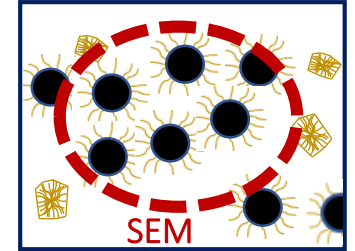
→ Spherulites $\sim 1\mu\text{m}$ diameter



Spherulite nucleus Inter-spherulitic zone

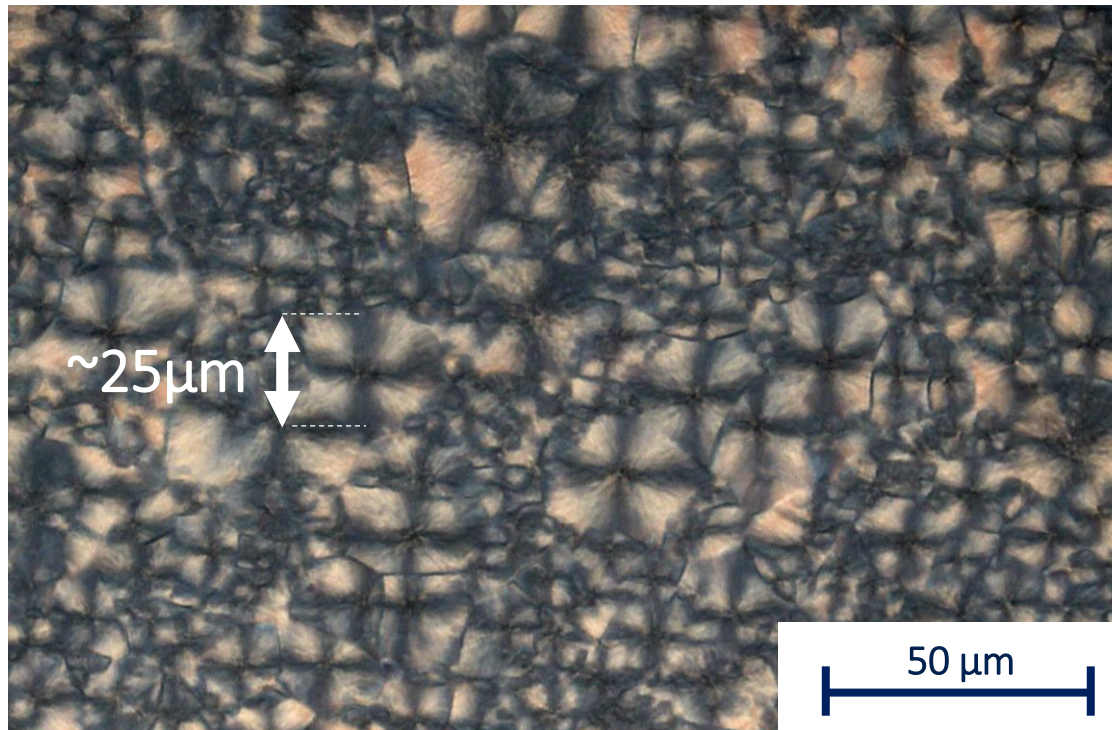
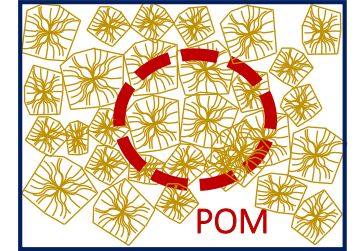
CHARACTERIZATION OF TC – Liq. N_2 fracture + SEM

└─→ Complicated in composite due to high V_{fiber}

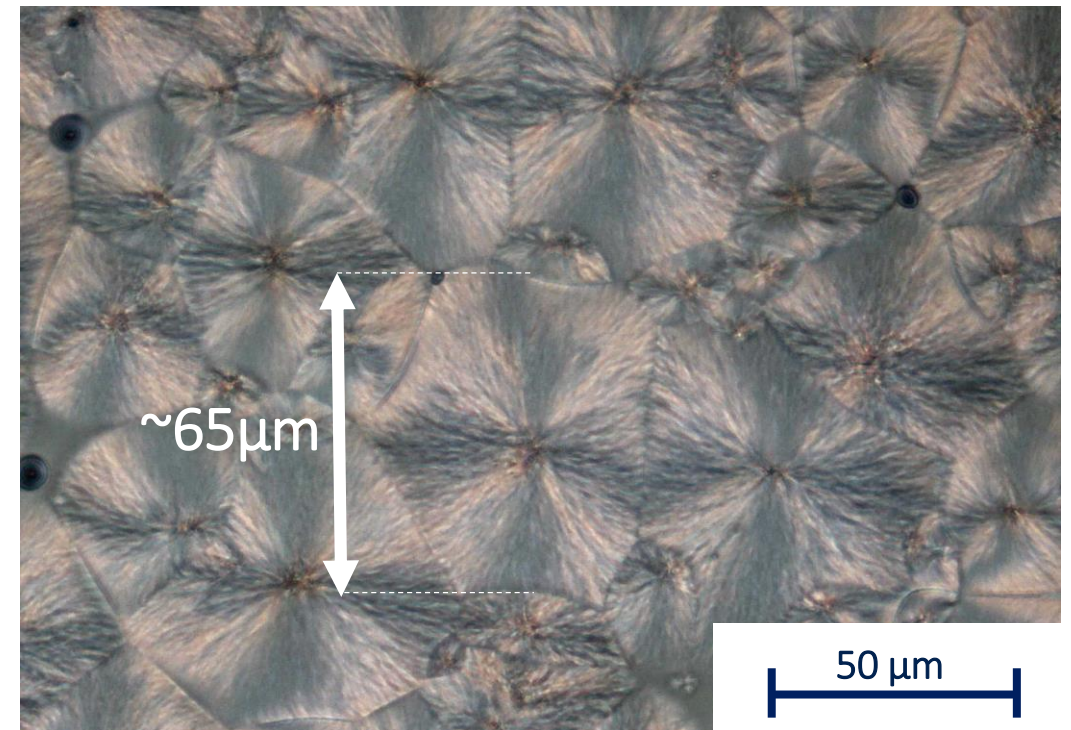


CHARACTERIZATION OF SPHERULITES – POM

└──► Suitable technique for spherulites observation



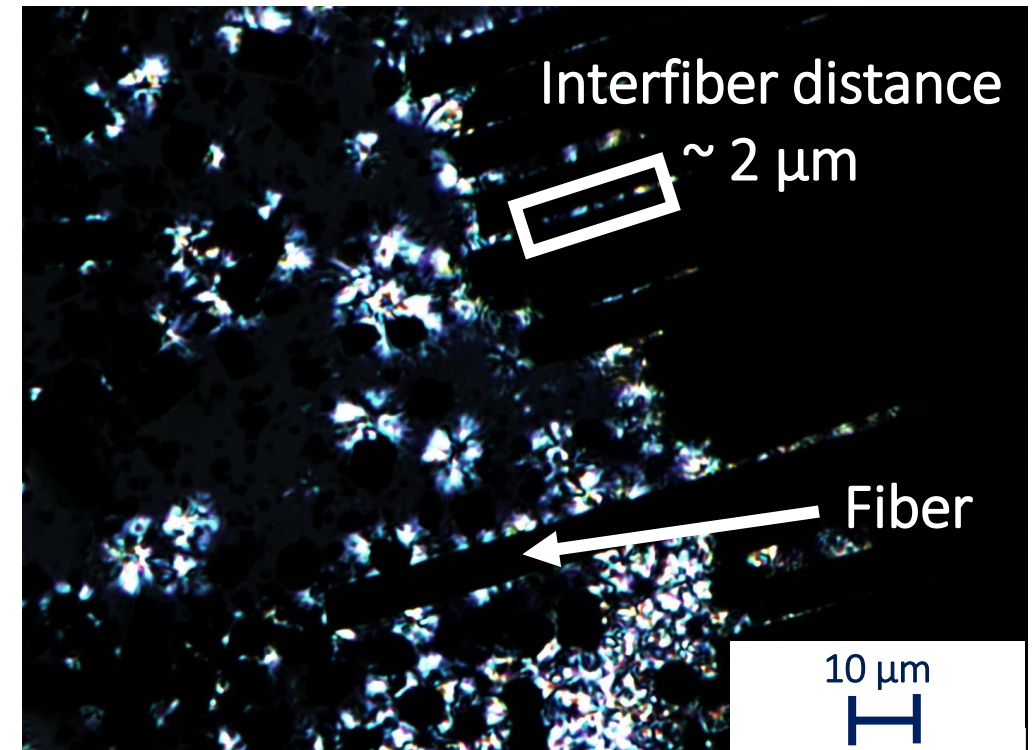
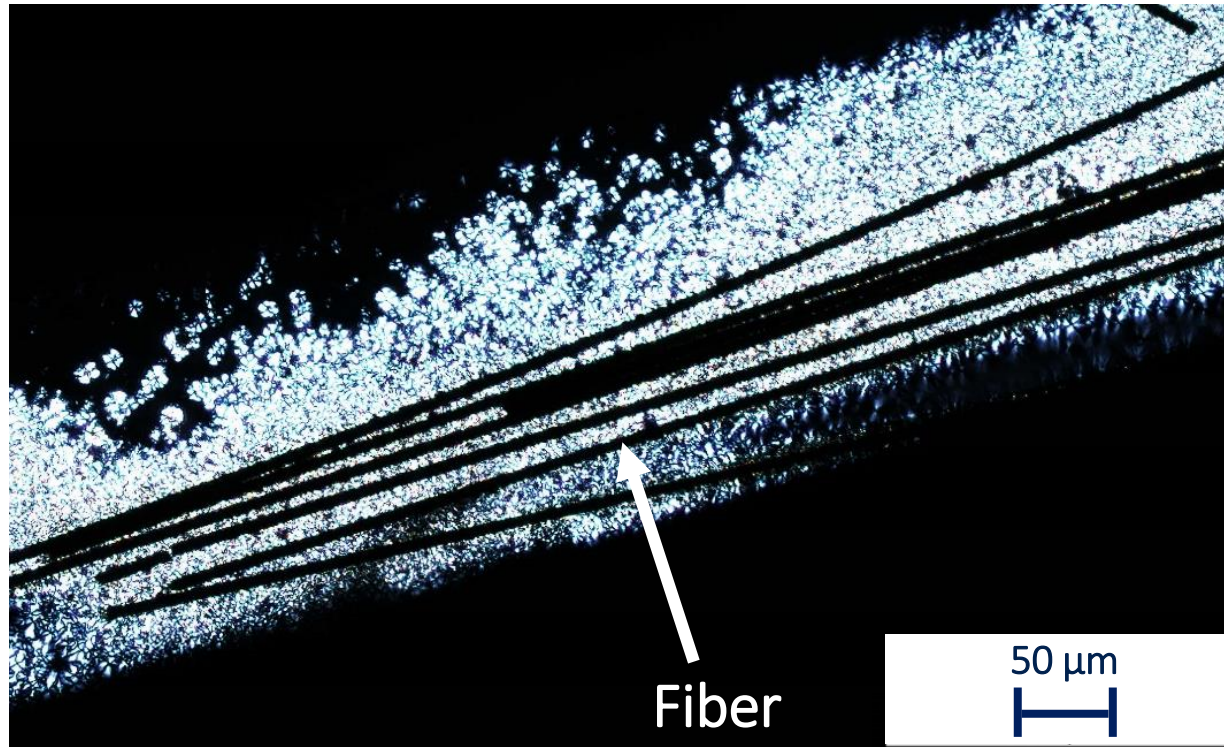
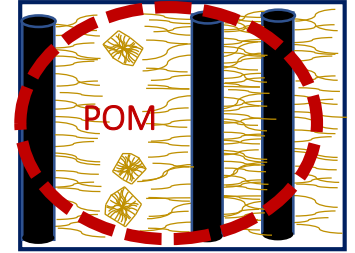
$T_{isoth. crist.} = 200^{\circ}\text{C}$



$T_{isoth. crist.} = 300^{\circ}\text{C}$

CHARACTERIZATION OF TC – POM

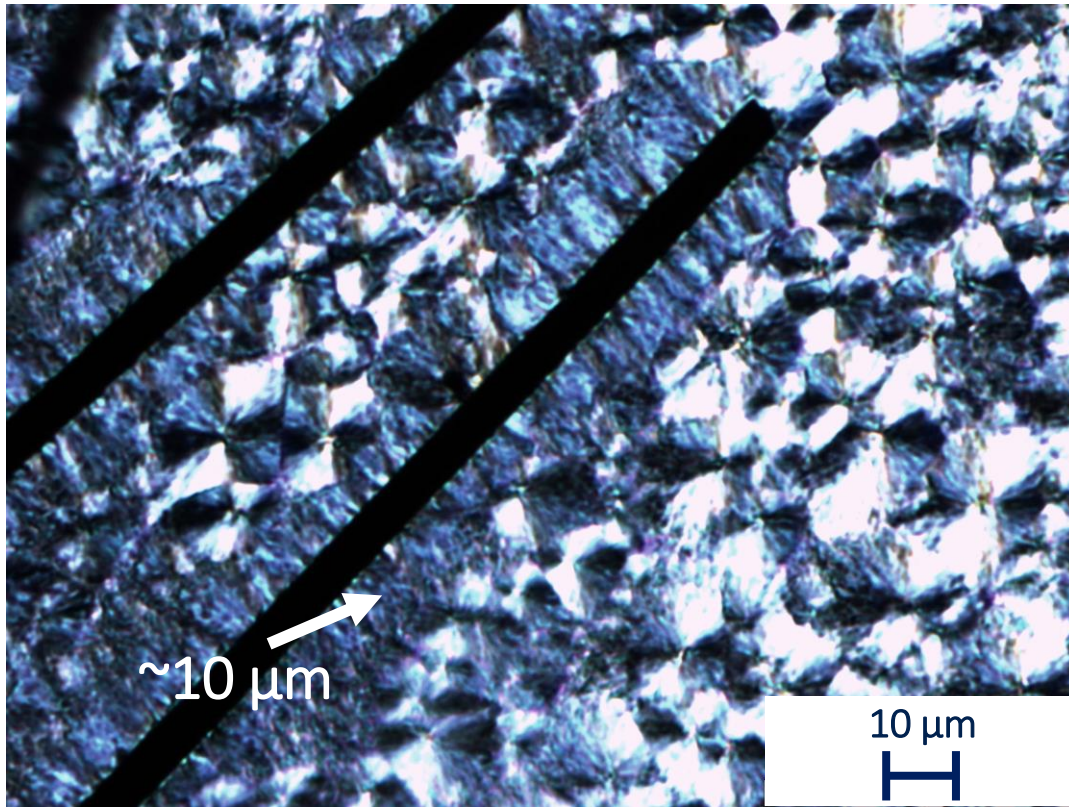
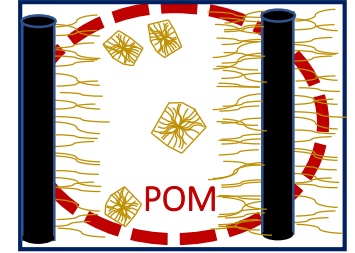
└─→ Complicated in composite due to high V_{fiber}



$$T_{isoth. crist.} = 315^{\circ}\text{C}$$

CHARACTERIZATION OF TC – POM

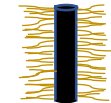
└─→ Easier in model samples with few fibers



$T_{isoth. crist.} = 200^{\circ}\text{C}$

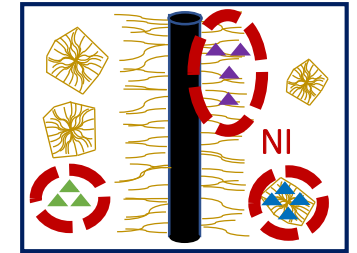


$T_{isoth. crist.} = 315^{\circ}\text{C}$

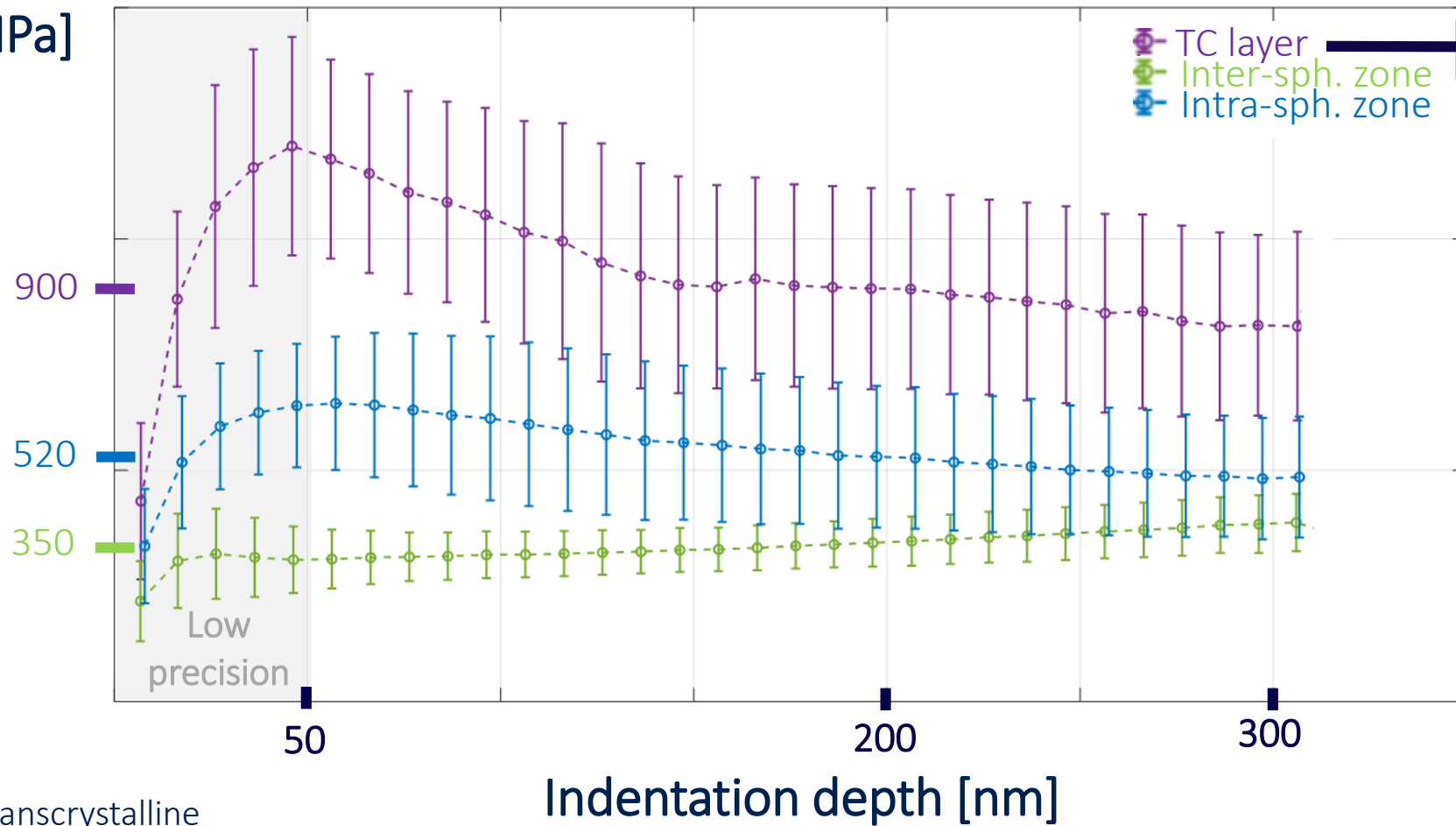


VARIATION OF THE LOCAL PEEK HARDNESS - NANOINDENTATION

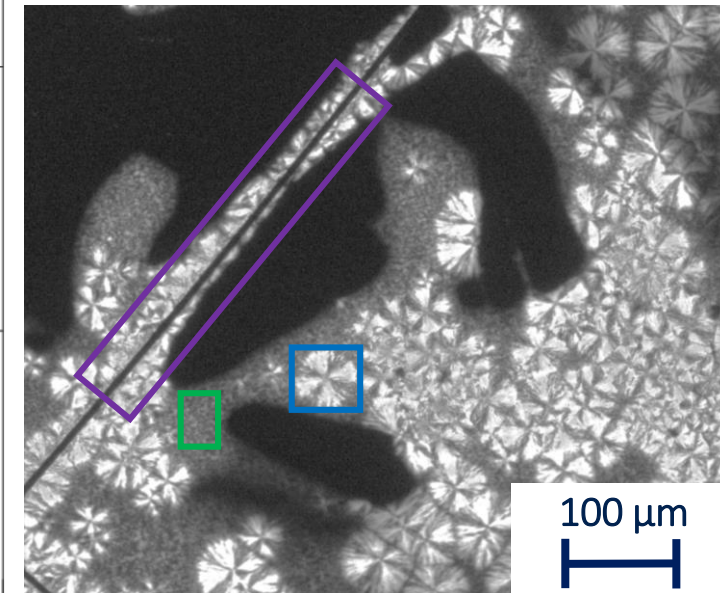
→ Hardness in intra-sph. zone and TC layer ~ 50 % and 150 % resp. higher than in inter-sph. zone (depth ~ 200 nm)



Nano Hardness
[MPa]

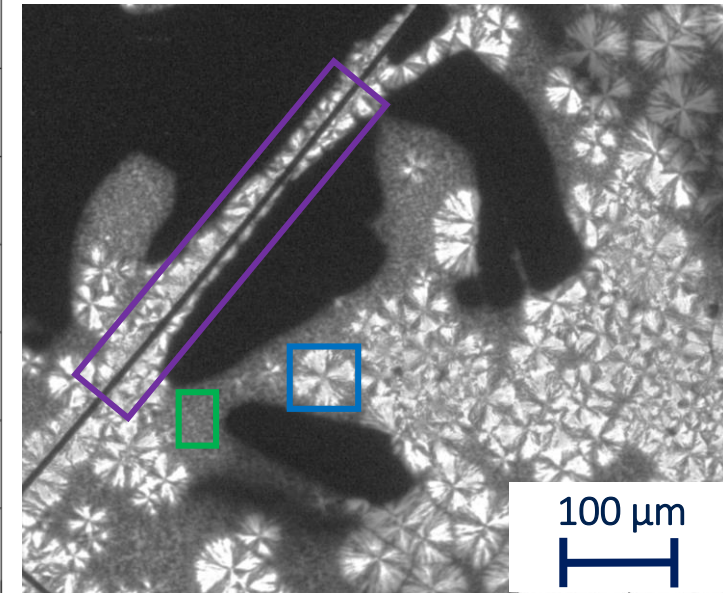
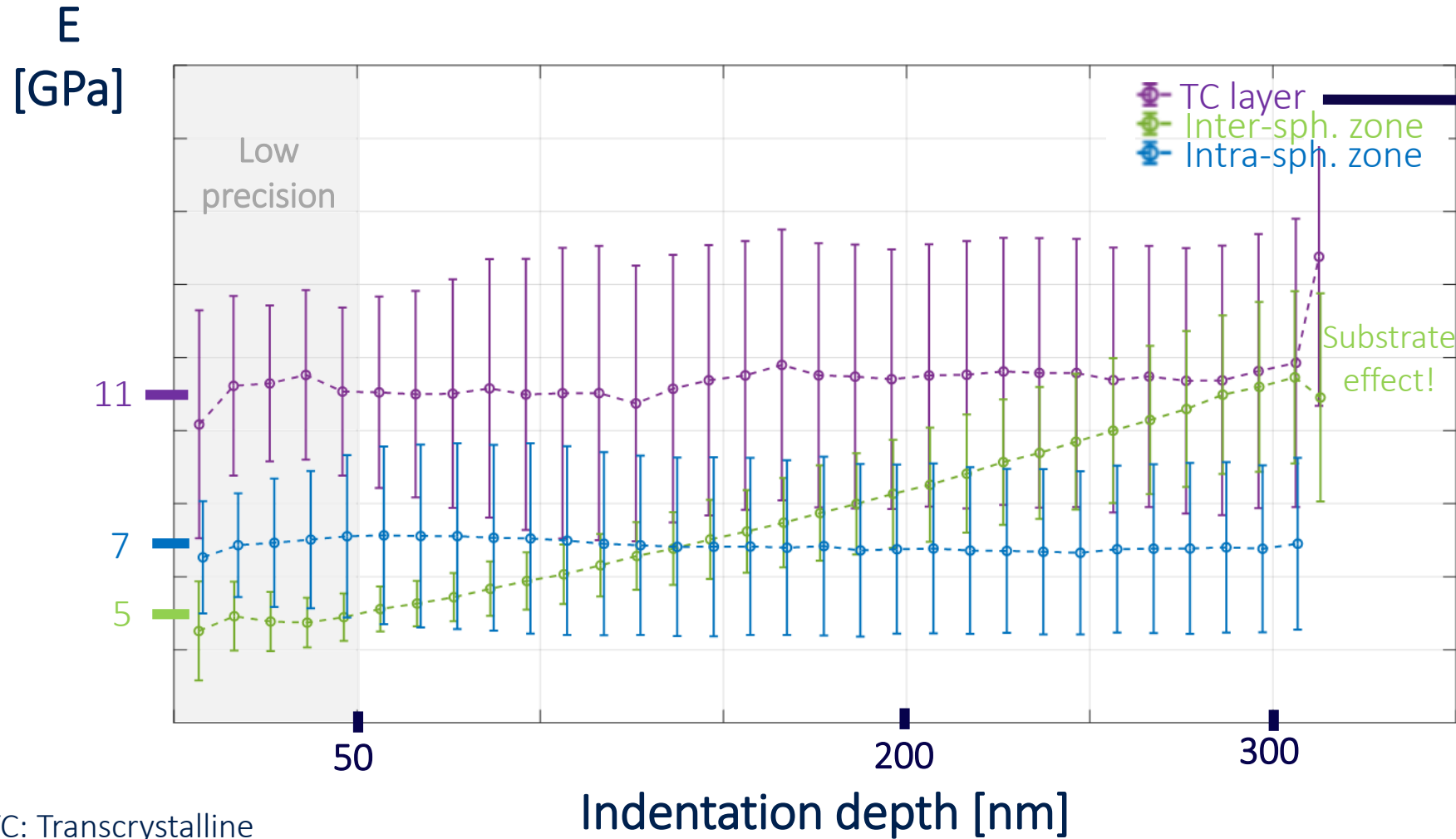
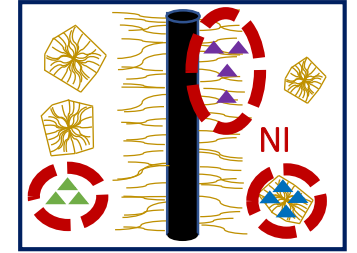


→ Indents at ~ 20 μm from the fiber



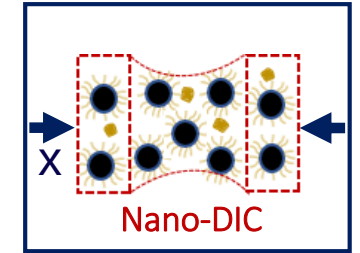
VARIATION OF THE LOCAL PEEK YOUNG'S MODULUS - NANOINDENTATION

→ E in TC layer ~ 70 % higher than in intra-sph. zone
(depth ~ 200 nm)

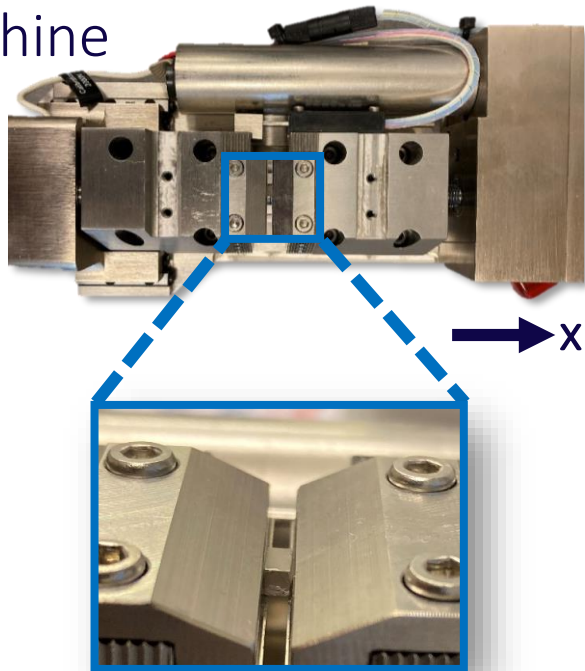


TRANSVERSE COMPRESSION TESTS IN SEM + NANO-DIC

└─ Access to local strain field in the interfiber zones showing intense plastic localization

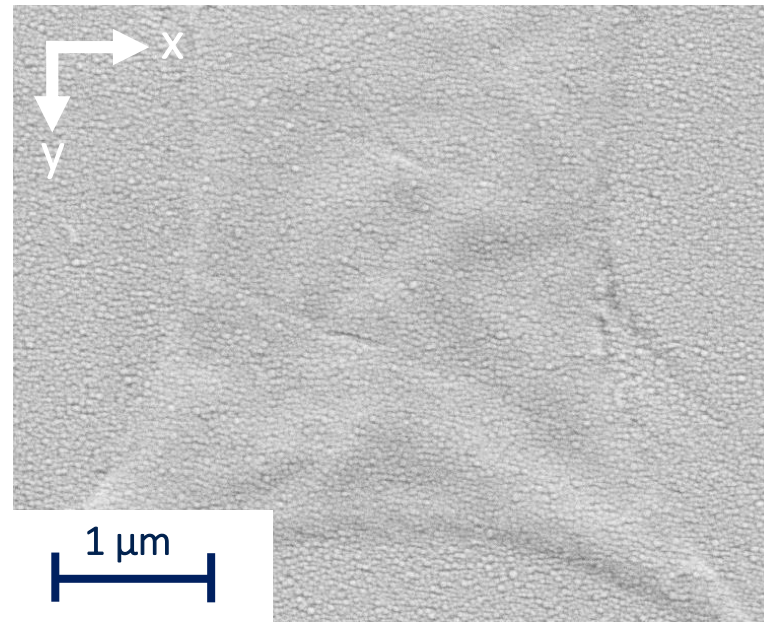


Mini compression machine

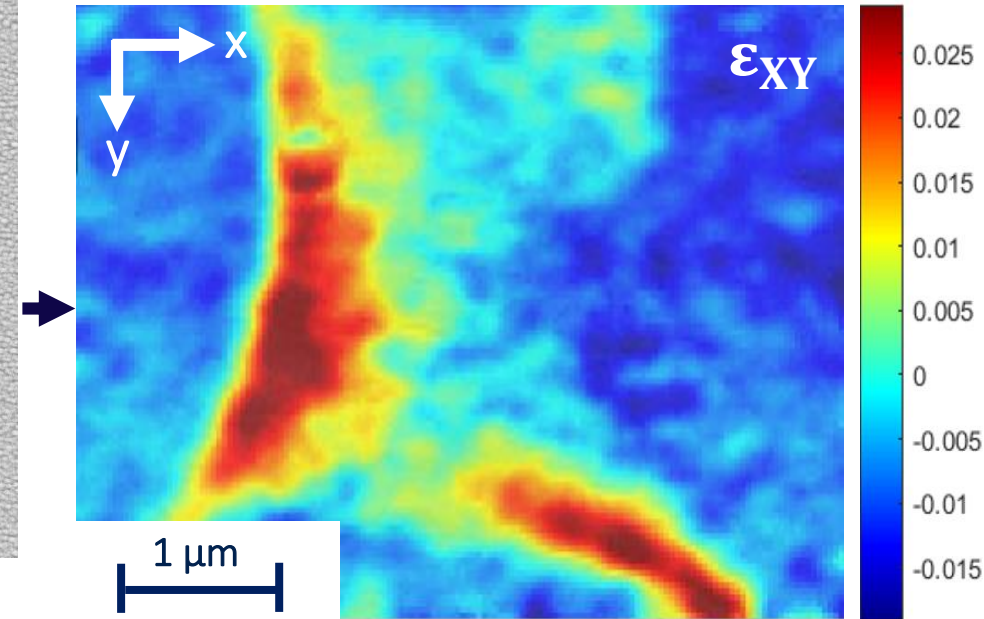


Unidirectional composite with nanometric speckle pattern

N. Klavzer, S. Gayot et al. (2023)



SEM imaging after each load increment



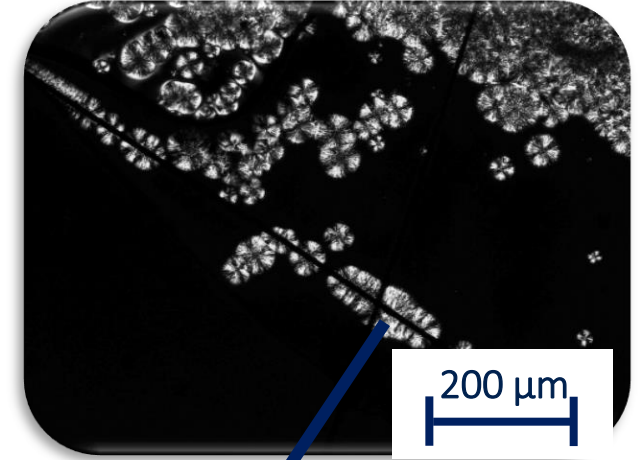
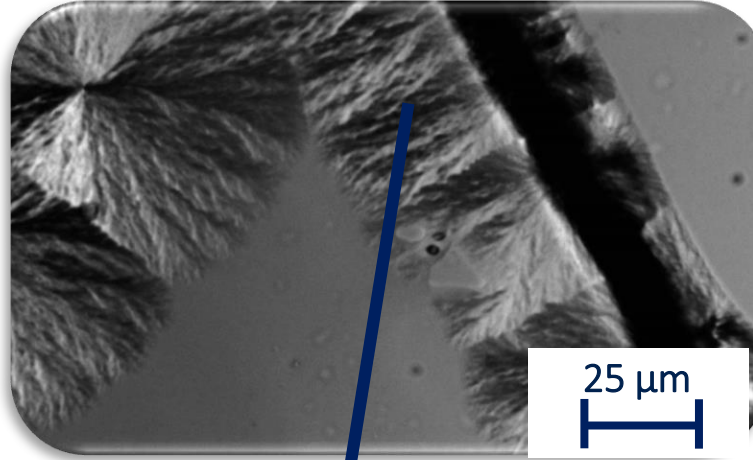
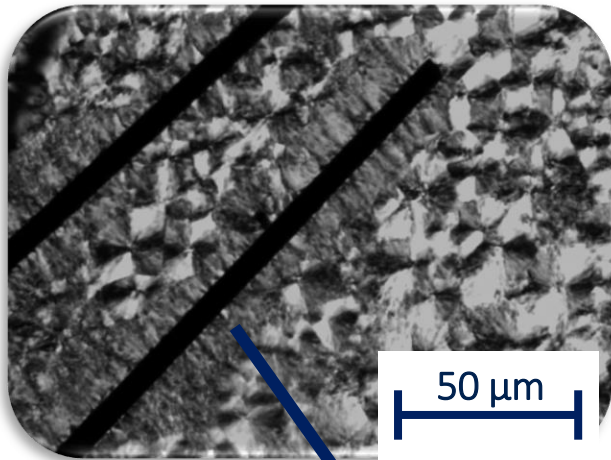
Micro-scale strain mapping from DIC

TAKE-HOME MESSAGES

PROCESSING
CONDITIONS

Semi-crystalline
←→
thermoplastic composite

PERFORMANCES

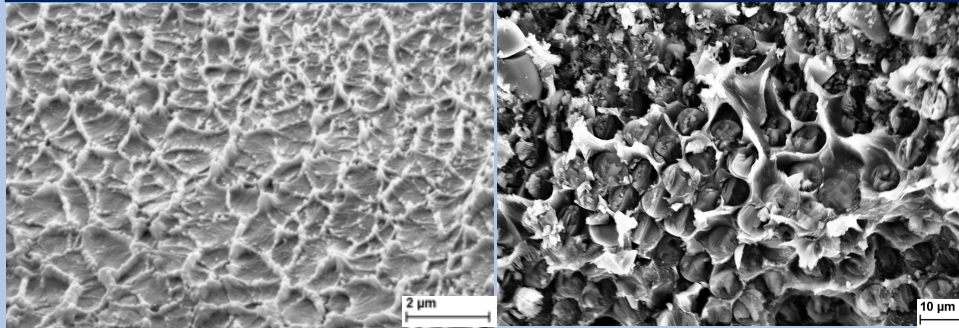


Impact on mechanical behavior in the interfiber zones ?

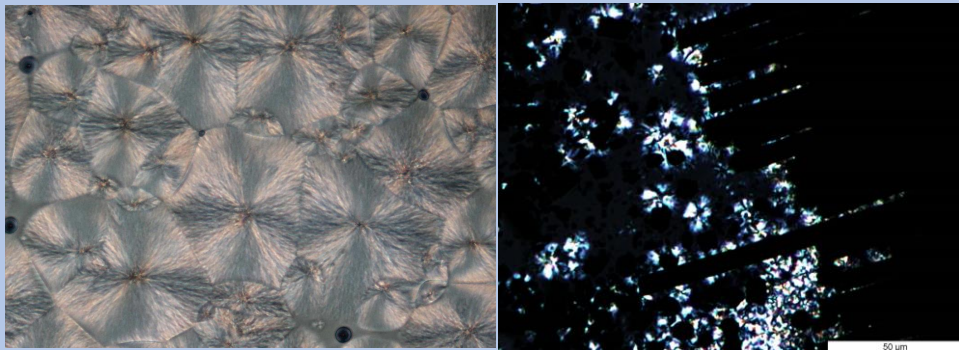
TAKE-HOME MESSAGES

MICROSTRUCTURE

Liq. N_2 fracture + SEM



POM



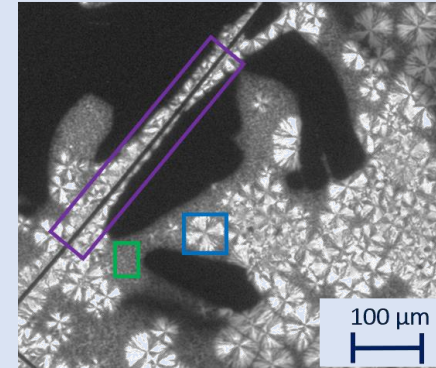
Characterization of PEEK microstructure in composite more complicated than in the neat PEEK due to small interfiber distances ($\sim 2 \mu\text{m}$)



Characterization on model PEEK/C fiber samples

NANO- /MICRO- MECHANICS

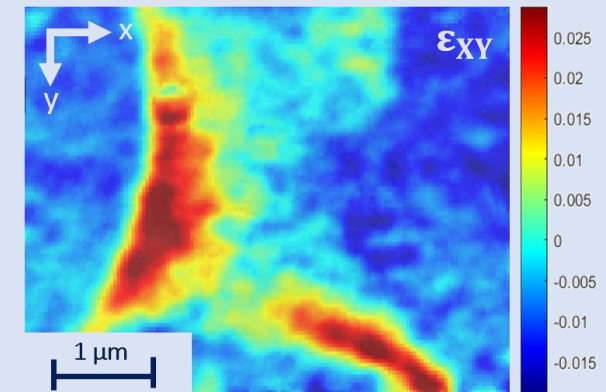
Nanoindentation



H in intra-sph. zone and TC layer $\sim 50\%$ and 150% resp. higher than in inter-sph. zone

E in TC layer $\sim 70\%$ higher than in intra-sph. zone

Transverse compression in SEM + Nano-DIC



Access to μm -scale strain field in the interfiber zones



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Back-up slides



Contents lists available at [ScienceDirect](#)

Composites Part A

journal homepage: www.elsevier.com/locate/compositesa



Nanoscale digital image correlation at elementary fibre/matrix level in polymer-based composites

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Thomas Pardoën^a

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Correlation Among Crystalline Morphology of PEEK, Interface Bond Strength, and In-Plane Mechanical Properties of Carbon/PEEK Composites

SHANG-LIN GAO, JANG-KYO KIM

Department of Mechanical Engineering, Hong Kong University of Science and Technology, Clear Water Bay, Hong Kong

Received 26 March 2001; accepted 29 May 2001

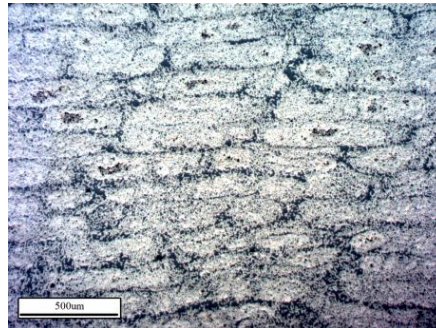
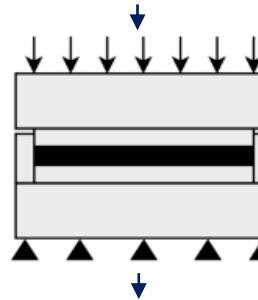
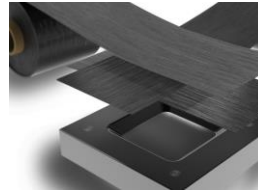
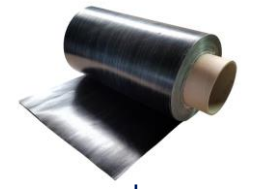
Table I Summary of the Tensile Test Results and Crystallinity of Neat PEEK Resin

Property	Cooling Rate (°C/min)					
	1	80	160	600	1000	2000
	Mean $\pm$ SD	Mean $\pm$ SD	Mean $\pm$ SD	Mean $\pm$ SD	Mean $\pm$ SD	Mean $\pm$ SD
Tensile strength (MPa)	108.5 $\pm$ 5.0	93.8 $\pm$ 9.0	92.5 $\pm$ 3.5	71.7 $\pm$ 10.4	57.5 $\pm$ 3.5	53.8 $\pm$ 3.3
Tensile modulus (GPa)	4.6 $\pm$ 0.5	—	4.0 $\pm$ 0.3	3.7 $\pm$ 0.1	3.3 $\pm$ 0.2	3.0 $\pm$ 0.2
Failure strain (%)	3.0 $\pm$ 1.4	9.8 $\pm$ 1.4	7.9 $\pm$ 3.0	—	115 $\pm$ 7.1	183.3 $\pm$ 5.8
Crystallinity (%)	38	30	28	26	19	17

MATERIALS



Semi-crystalline
thermoplastic

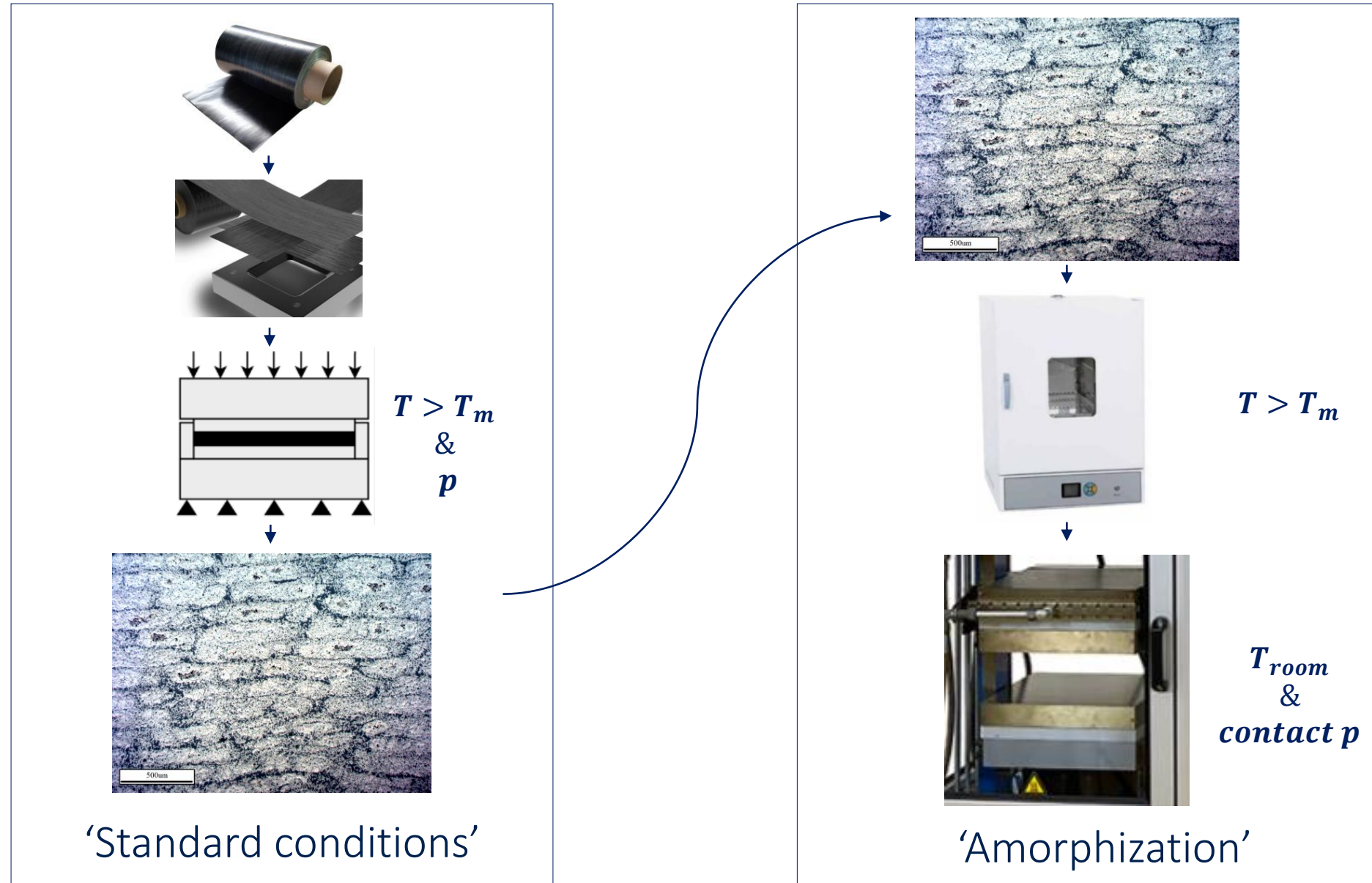


Unidirectional (UD)
composite

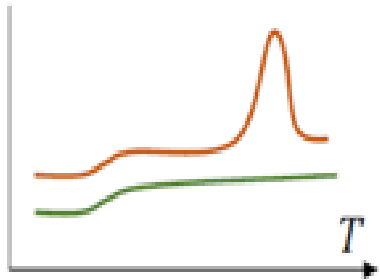


'Model'
samples

UD COMPOSITE



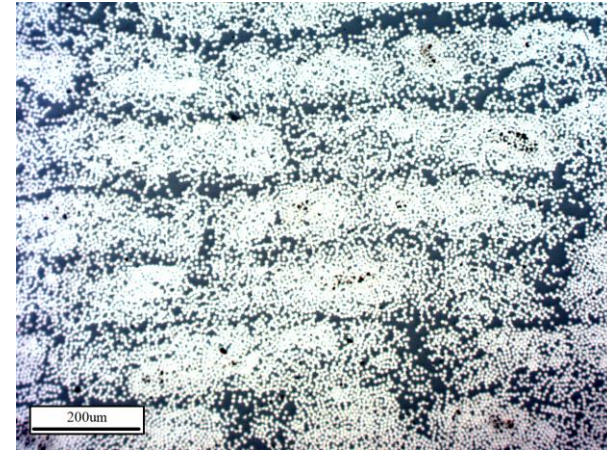
DEGREE OF CRYSTALLINITY - χ_c



DSC



‘Standard conditions’

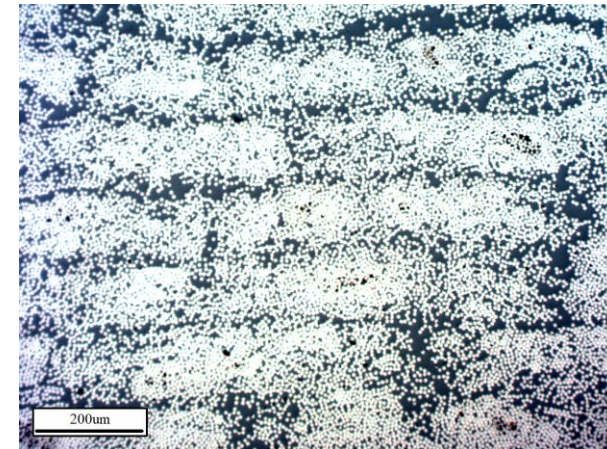


$\chi_c \sim 34 \%$

$\chi_c \sim 36 \%$

$\chi_c \sim 28 \%$

‘Amorphization’



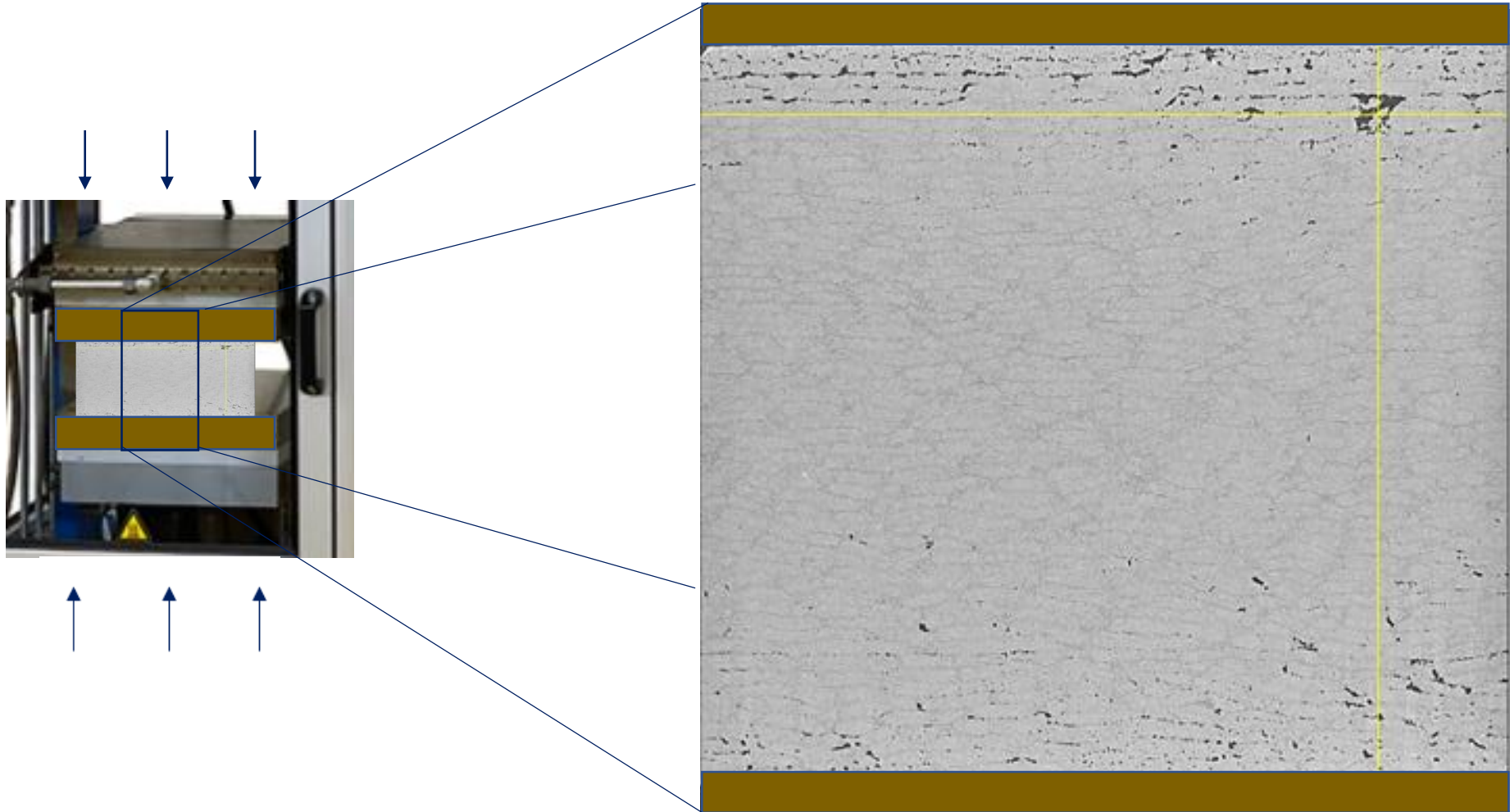
$\chi_c \sim 15 \%$

$\chi_c \sim 28 \%$

$\chi_c \sim 18 \%$



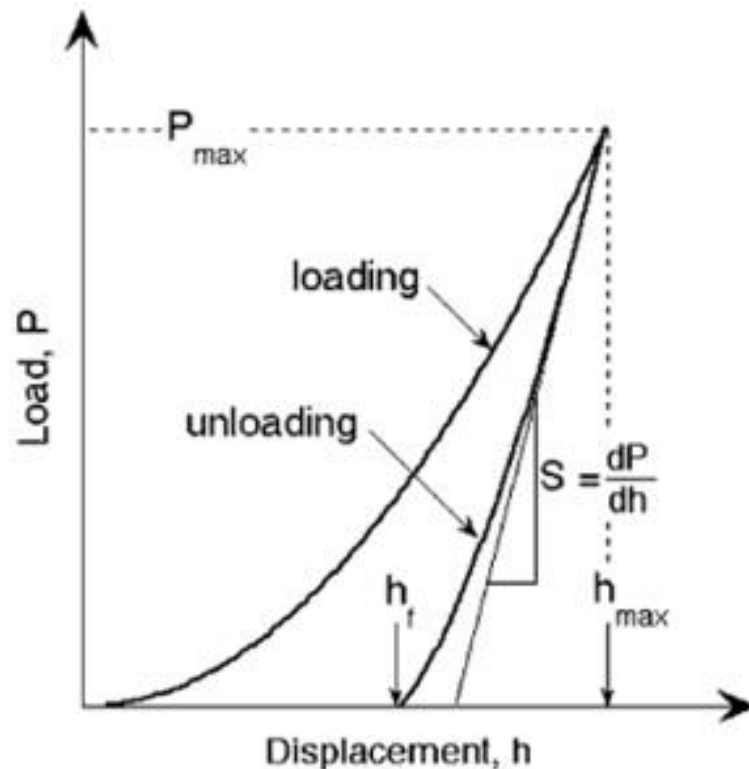
DEFECTS DETECTION – TOMOGRAPHY



PRINCIPLE OF NANOINDENTATION

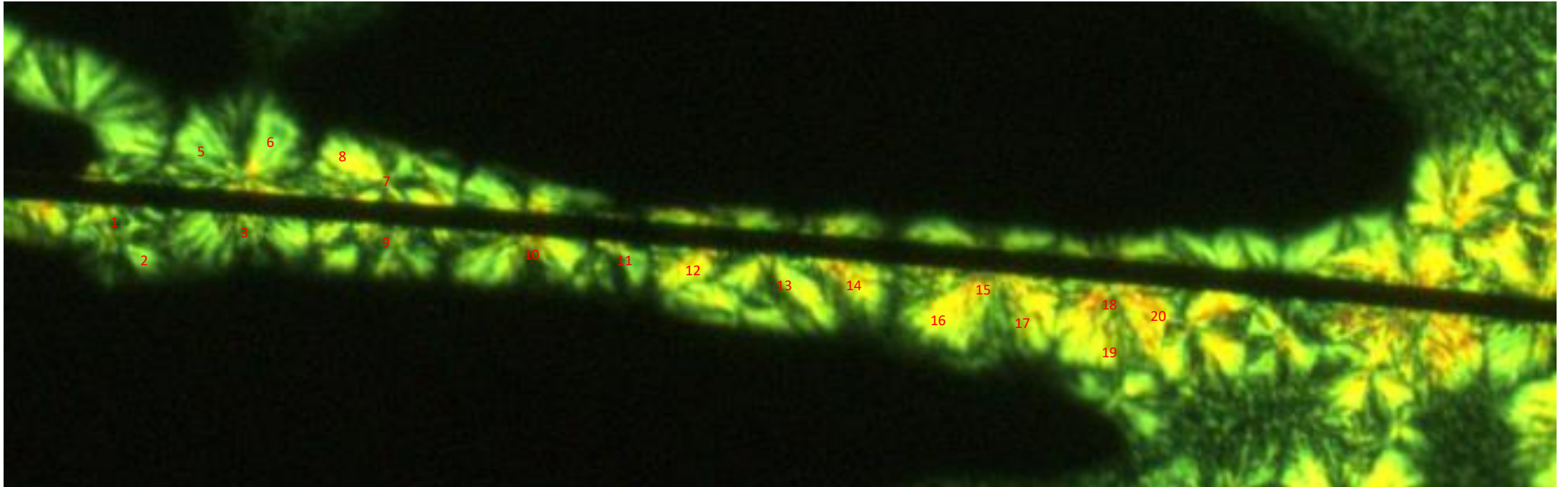
Instrumented nanoindentation: *indentation depth* of the tip and the *force* are measured continuously during the test

Direct calculation of *Young modulus* and *hardness* from the “*force vs depth*” curve *



$$H = \frac{P_{max}}{A}$$
$$S = \beta \frac{2}{\sqrt{\pi}} E_{eff} \sqrt{A}$$
$$\frac{1}{E_{eff}} = \frac{1 - \nu^2}{E} + \frac{1 - \nu_i^2}{E_i}$$

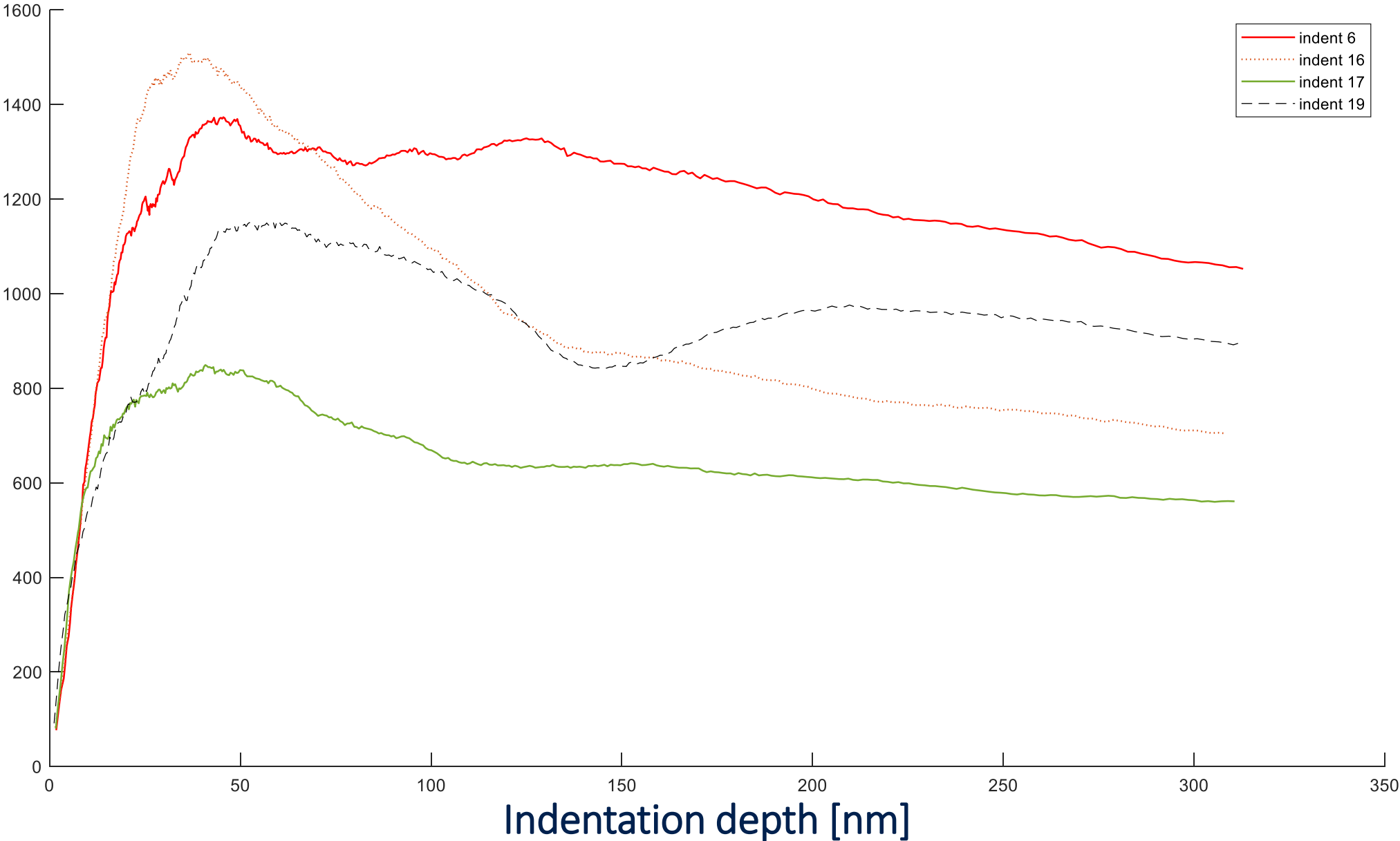
VARIATION OF THE LOCAL PEEK HARDNESS - NANOINDENTATION



TC layer

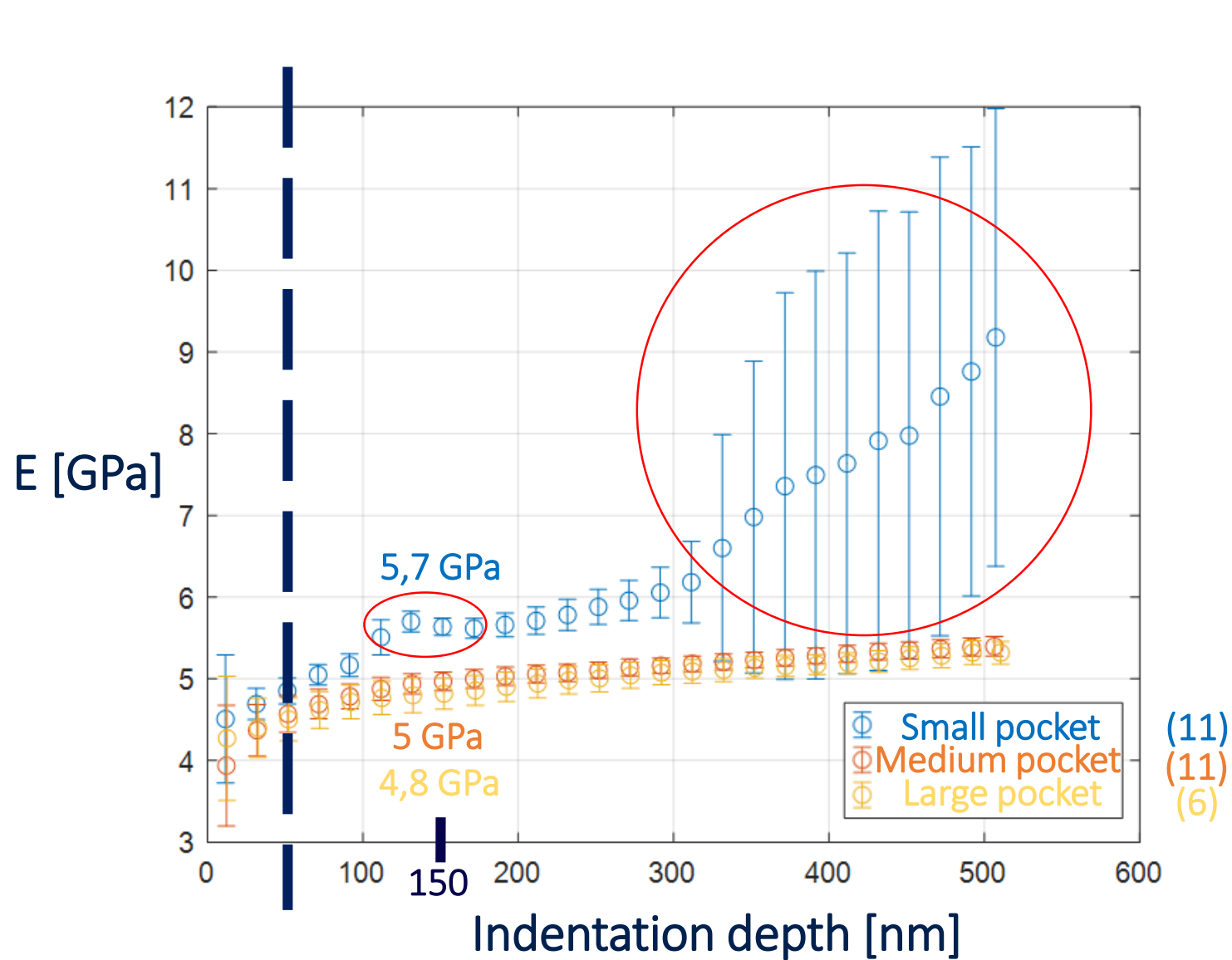
VARIATION OF THE LOCAL PEEK HARDNESS - NANOINDENTATION

Nano
hardness
[MPa]

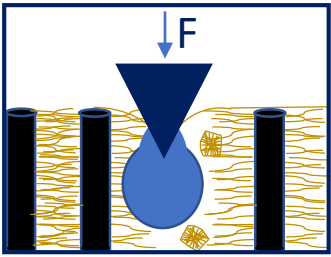
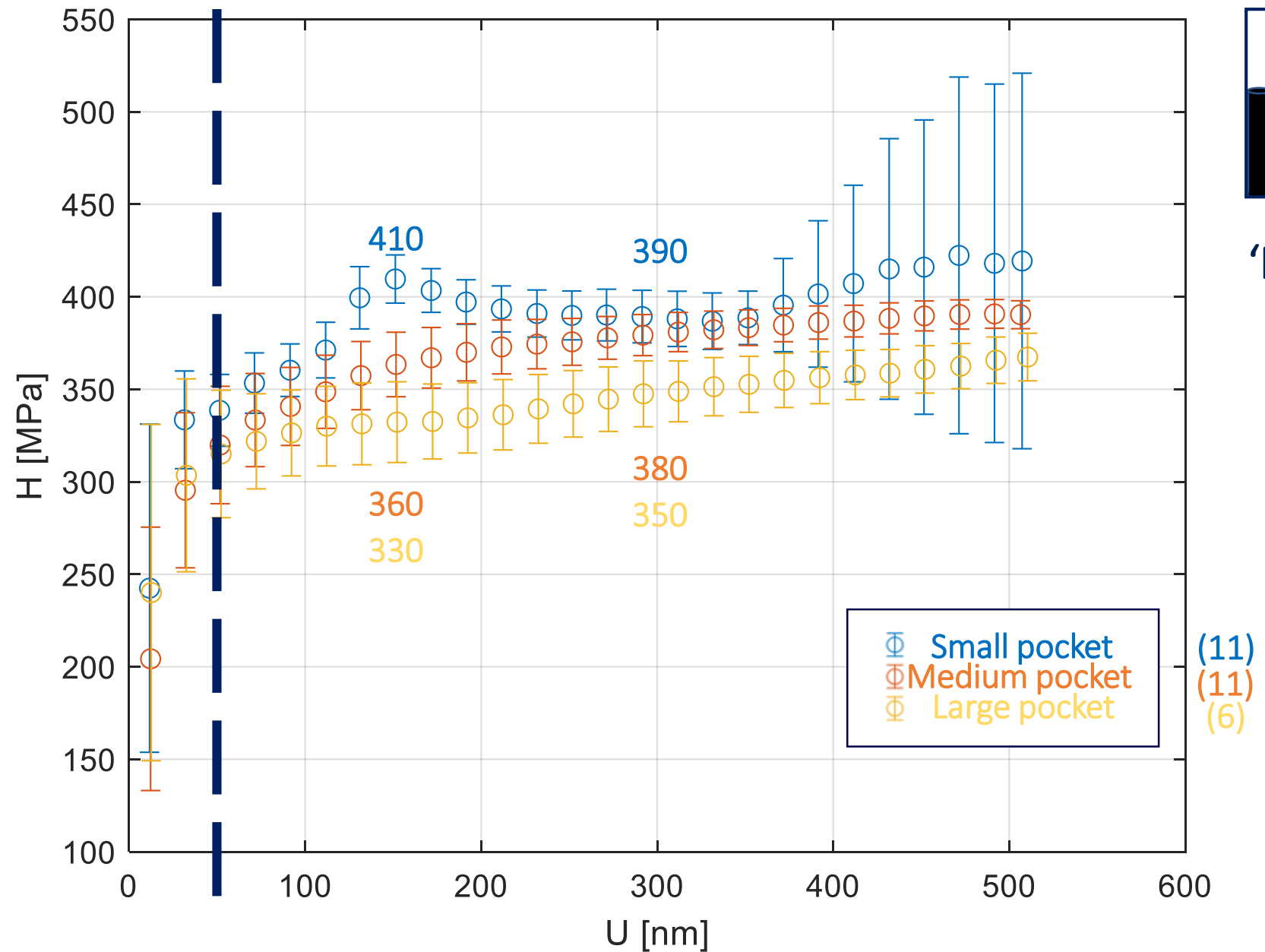


~17 μm
~20 μm
~17 μm
~23 μm

NEAR FIBER MATRIX MECHANICAL PROPERTIES - NANOINDENTATION



NEAR FIBER MATRIX MECHANICAL PROPERTIES - NANOINDENTATION

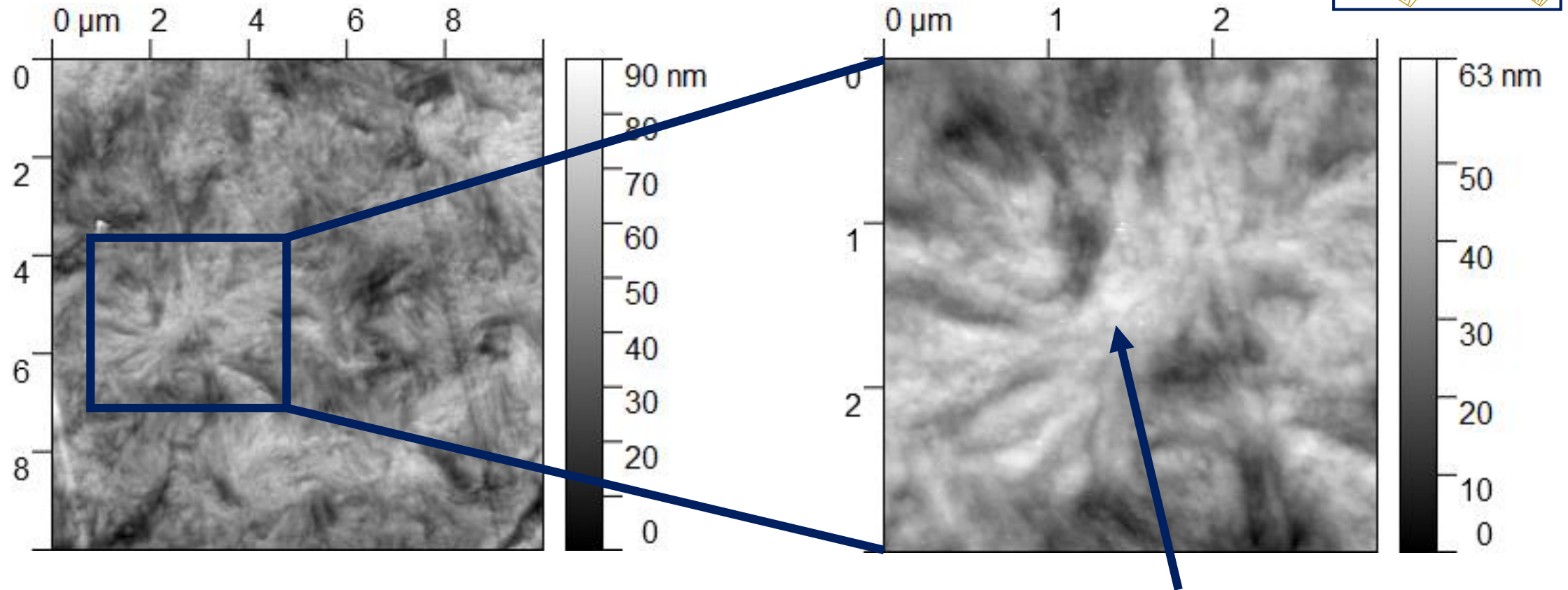


‘PEEK pocket’



CHARACTERIZATION OF SPHERULITES- AFM

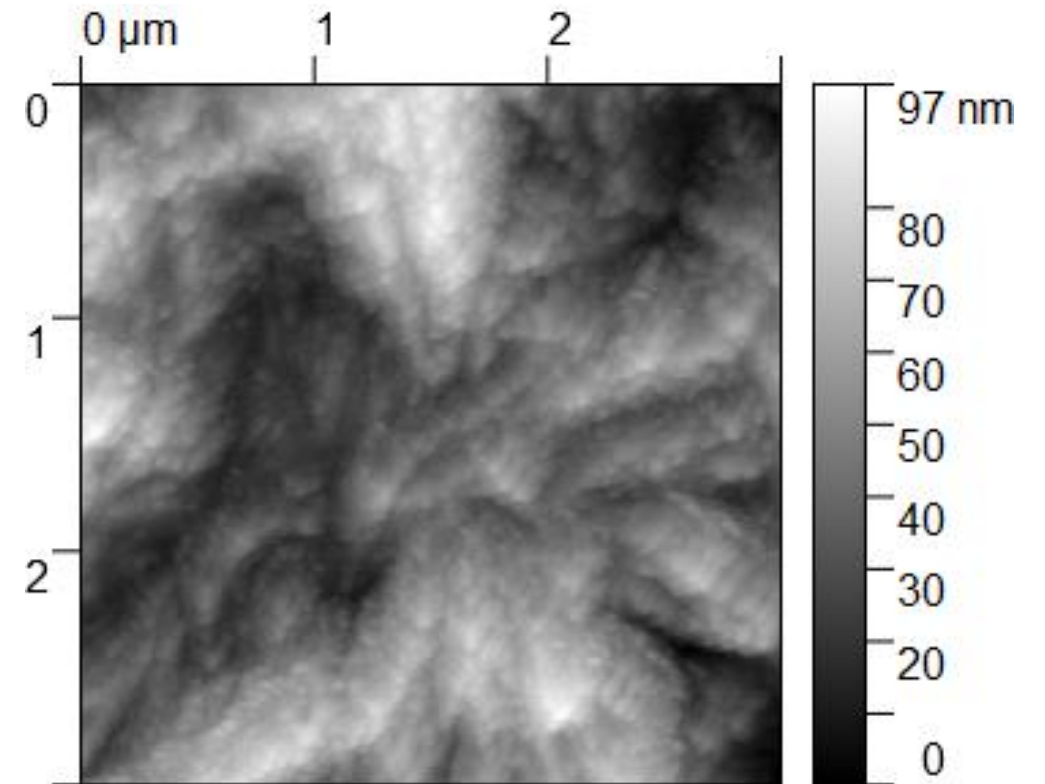
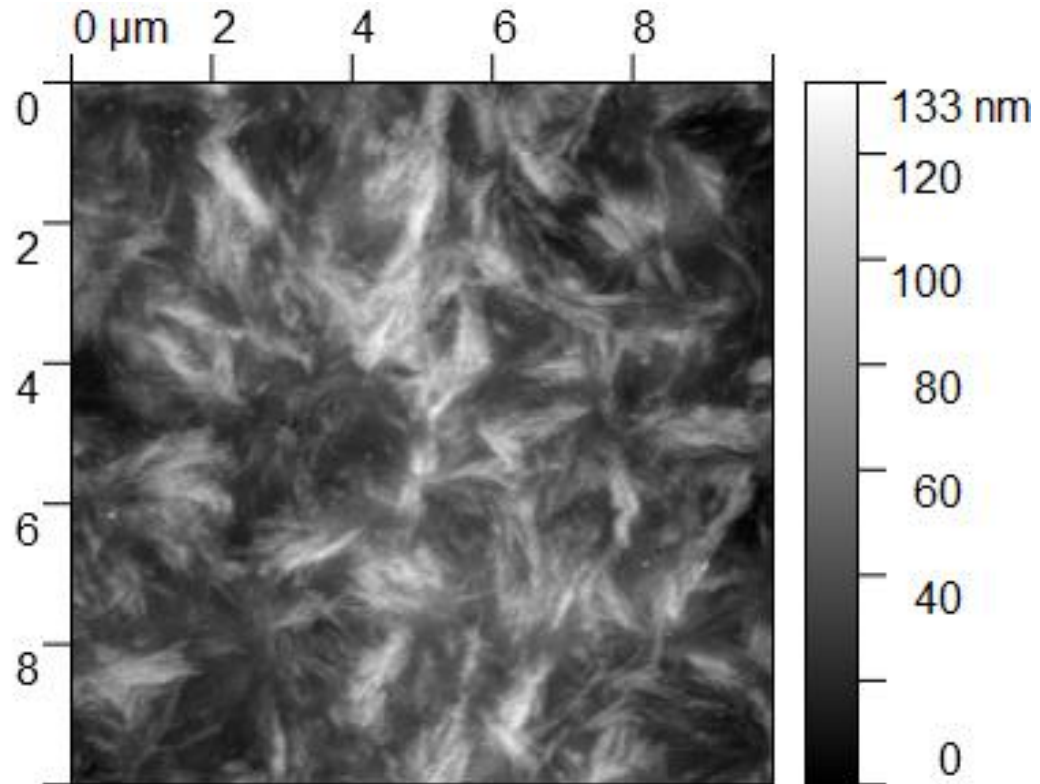
Topographic contrast



Spherulite nucleus

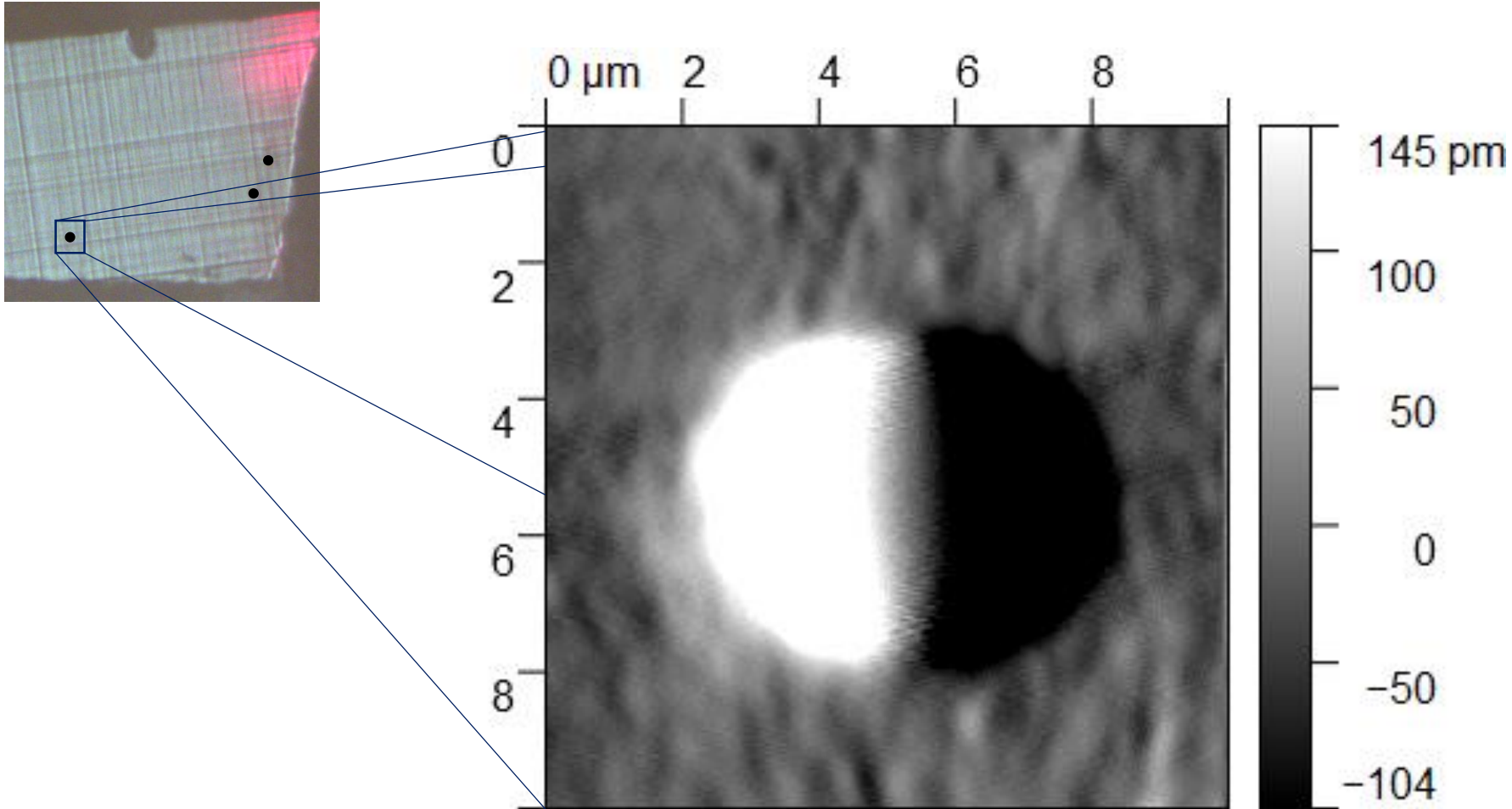
SPHERULITES OBSERVATIONS - AFM

Topographic contrast
Etched samples



TRANSCRYSTALIZATION OBSERVATIONS - AFM

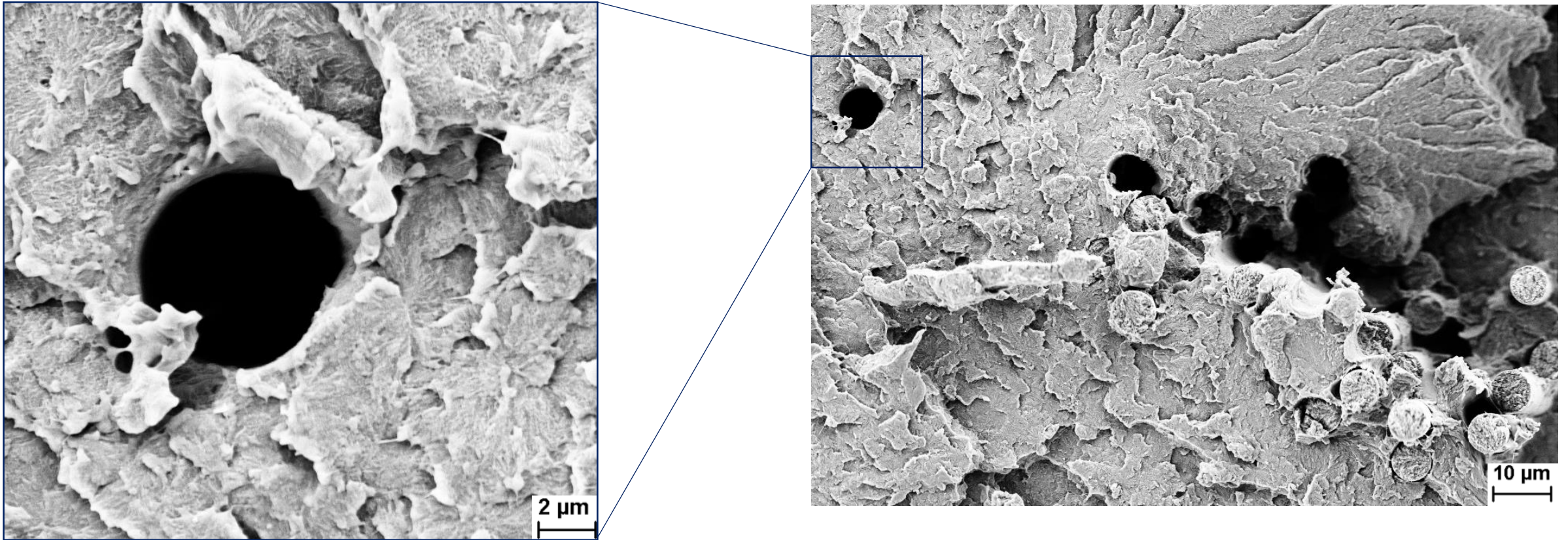
Error image



Difficulties:

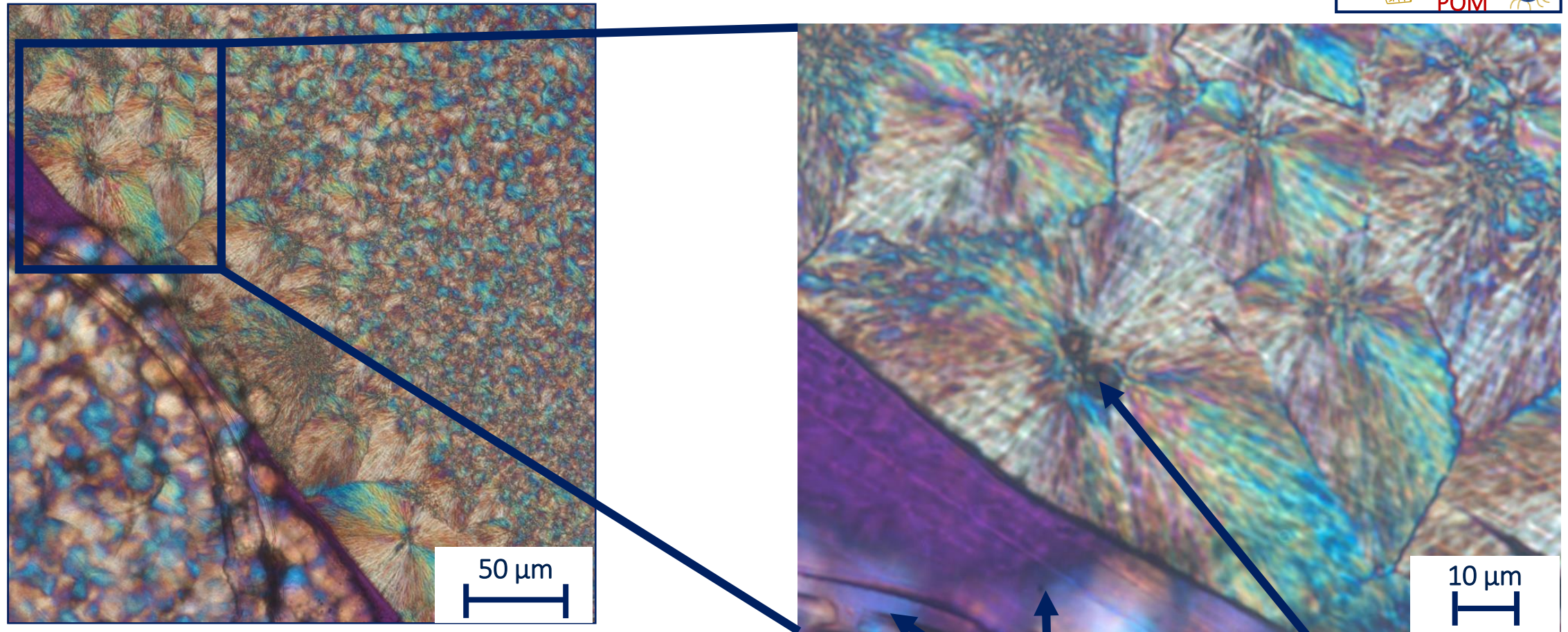
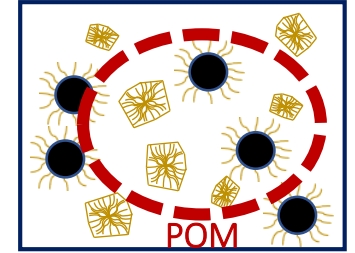
- Differential erosion
- Fiber/matrix interface decohesion during sample preparation
- Presence of fibers => cutting line on the surface

TRANSCRYSTALIZATION OBSERVATIONS - Liq. N_2 fracturing + SEM



‘Model’ samples

TRANSCRYSTALLIZATION OBSERVATIONS - POM

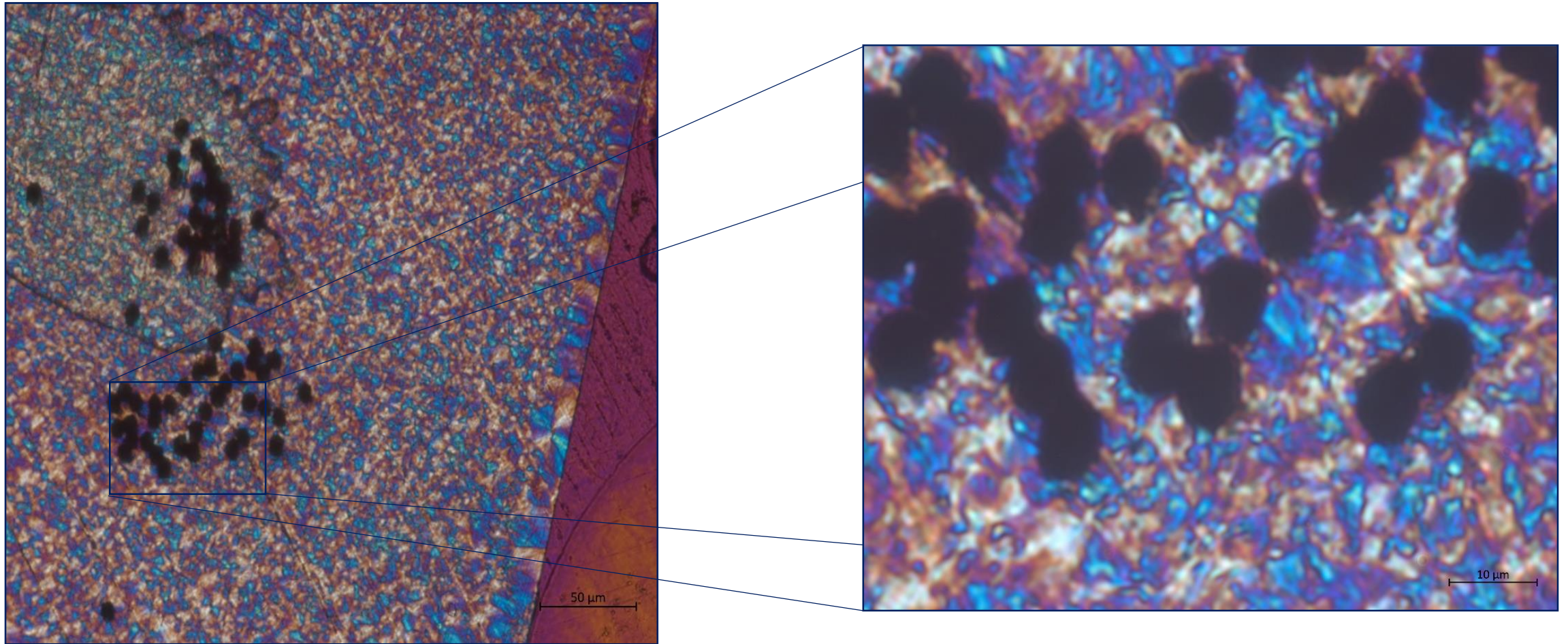


POM: polarized optical microscopy

Sample holder

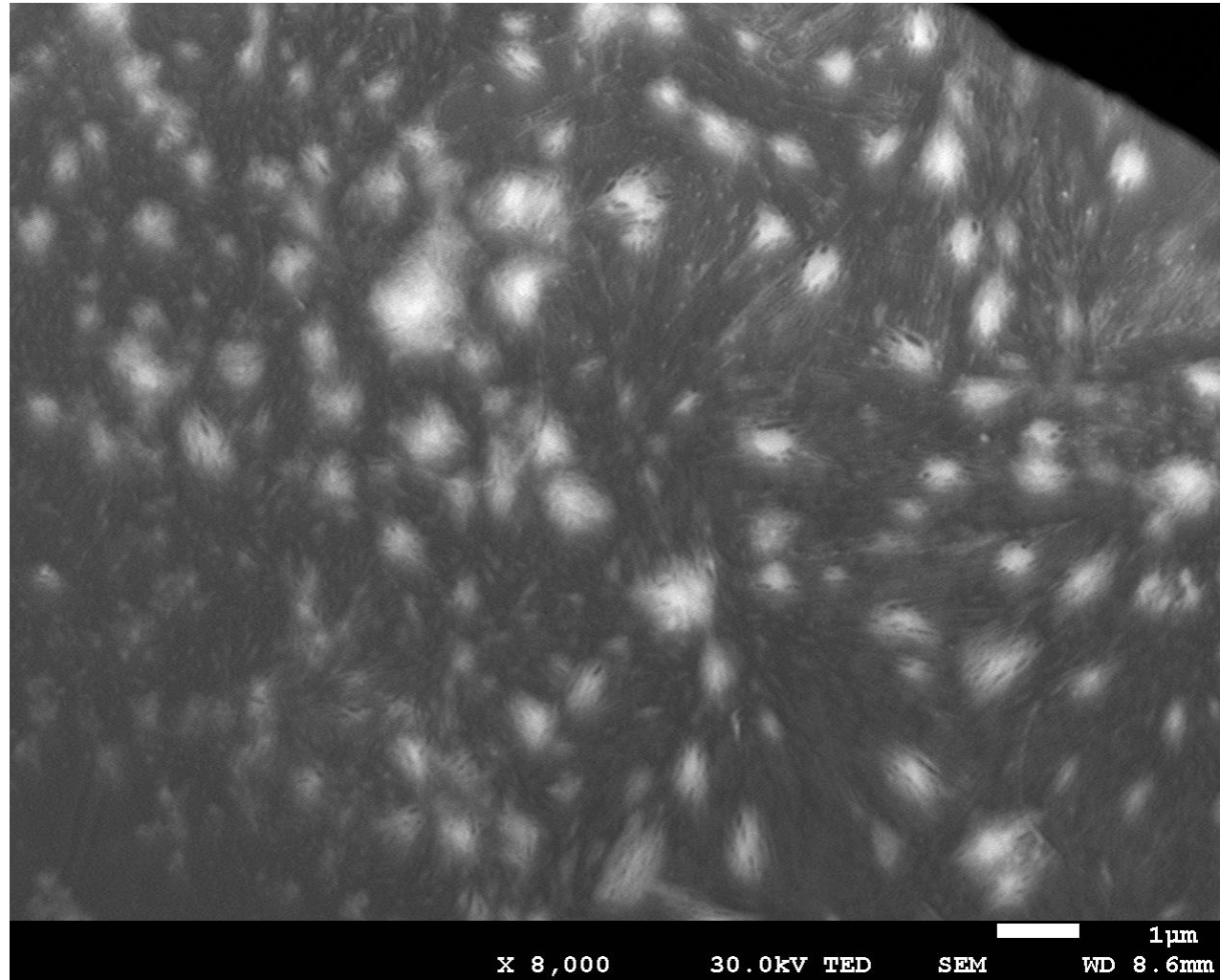
Fibers

TRANSCRYSTALIZATION OBSERVATIONS - POM



‘Model’ samples

SPHERULITES OBSERVATIONS - STEM



SPHERULITES OBSERVATIONS - PERMANGANATIC ETCHING + SEM

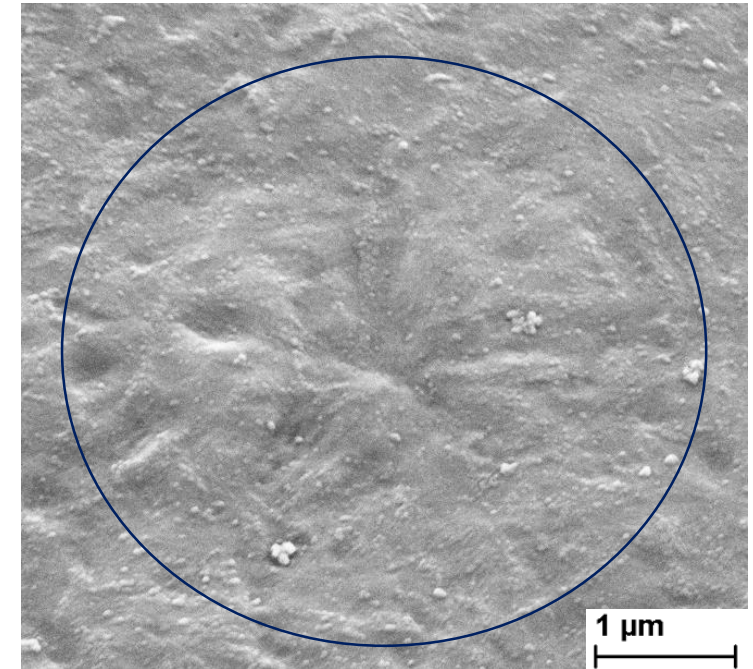
Chemical etching step (5 min):
sulfuric acid, phosphoric acid, distilled
water & potassium permanganate

Rinsing step (1 min):
hydrogen peroxyde & water

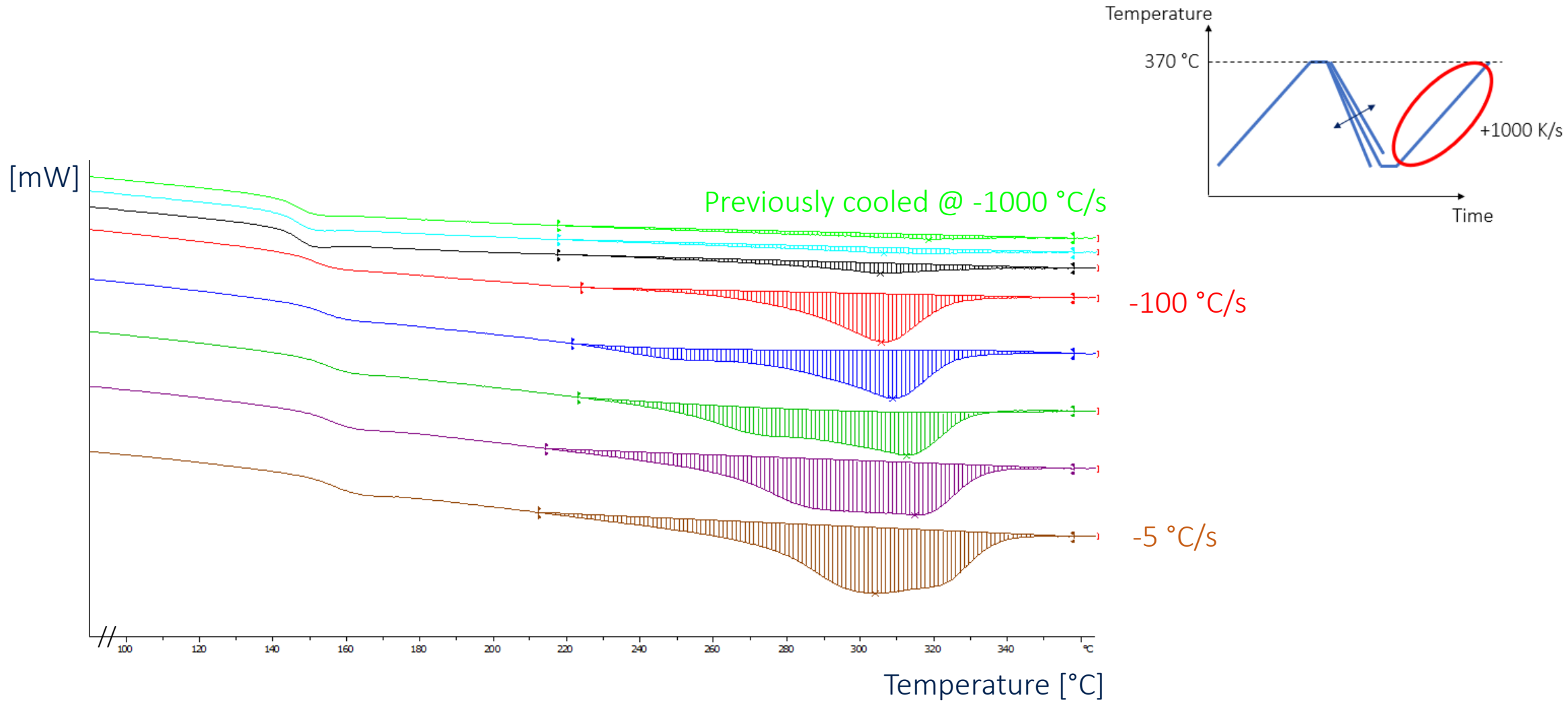
Drying step:
Acetone



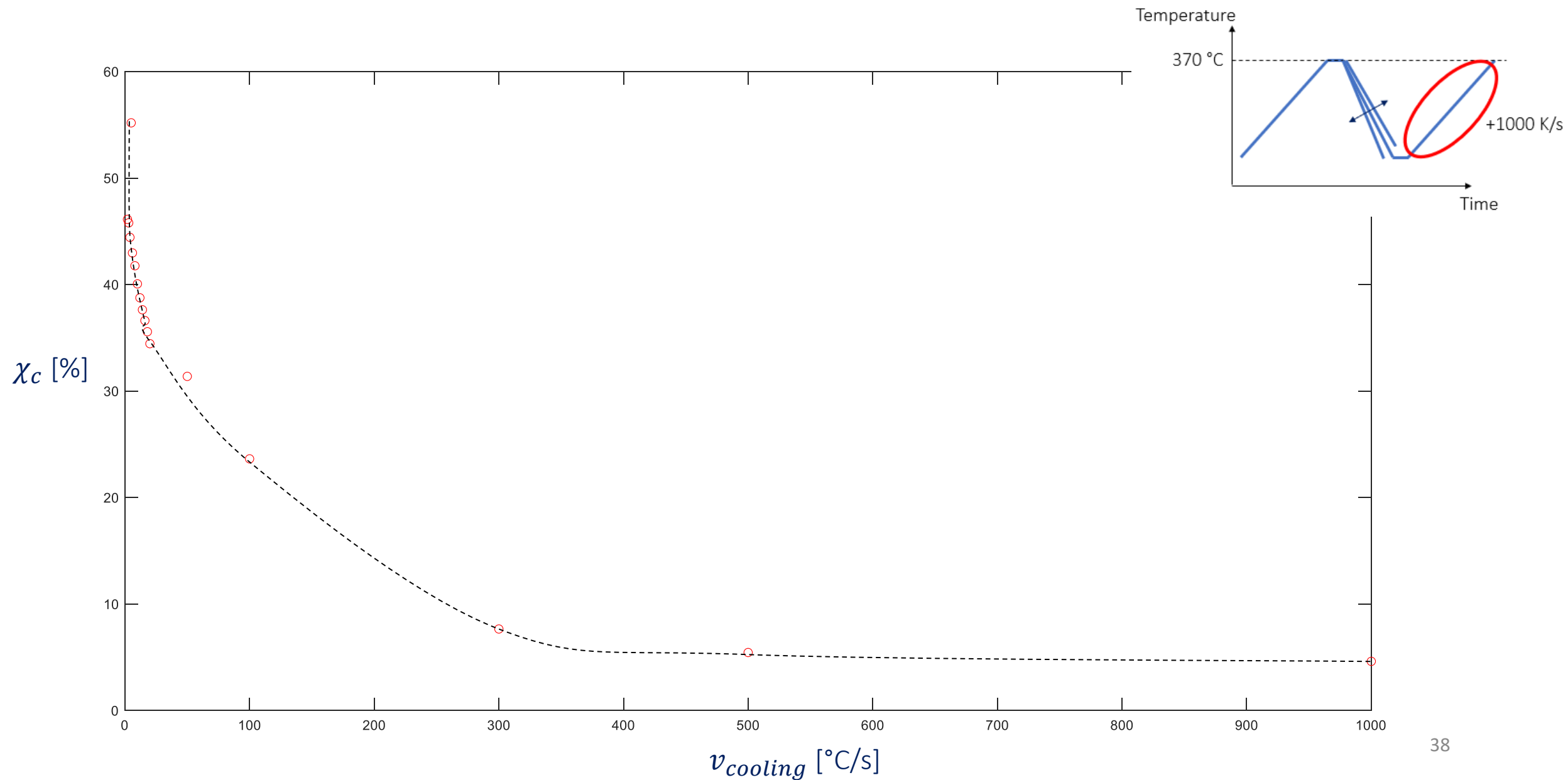
M. Debaut (1990-1991)



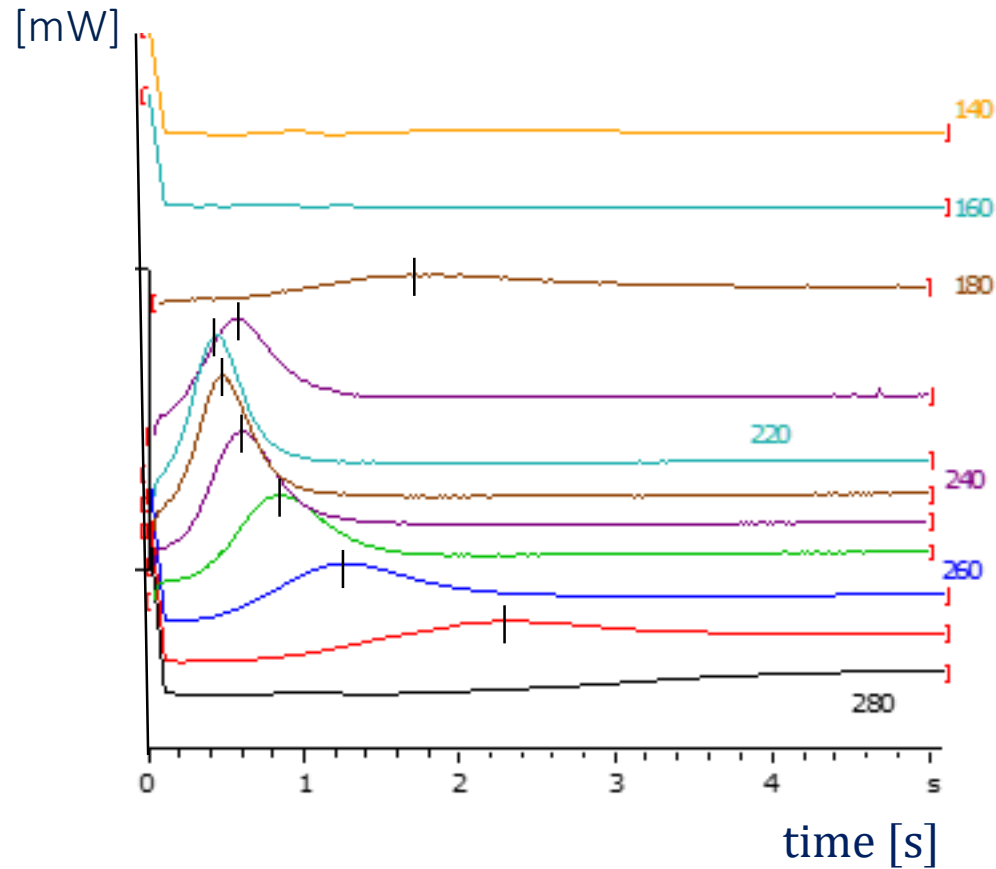
DEGREE OF CRYSTALLINITY – IMPACT OF COOLING RATE



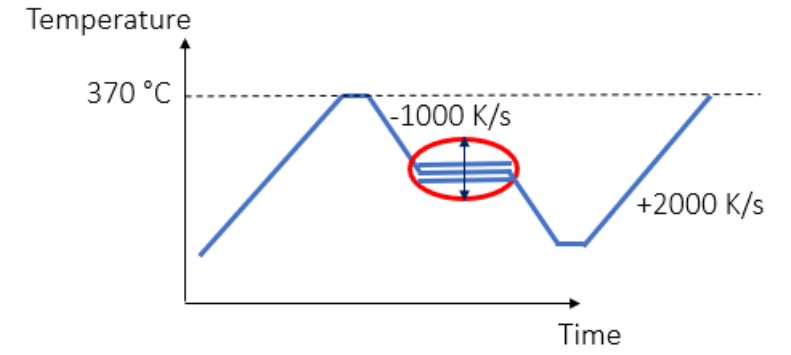
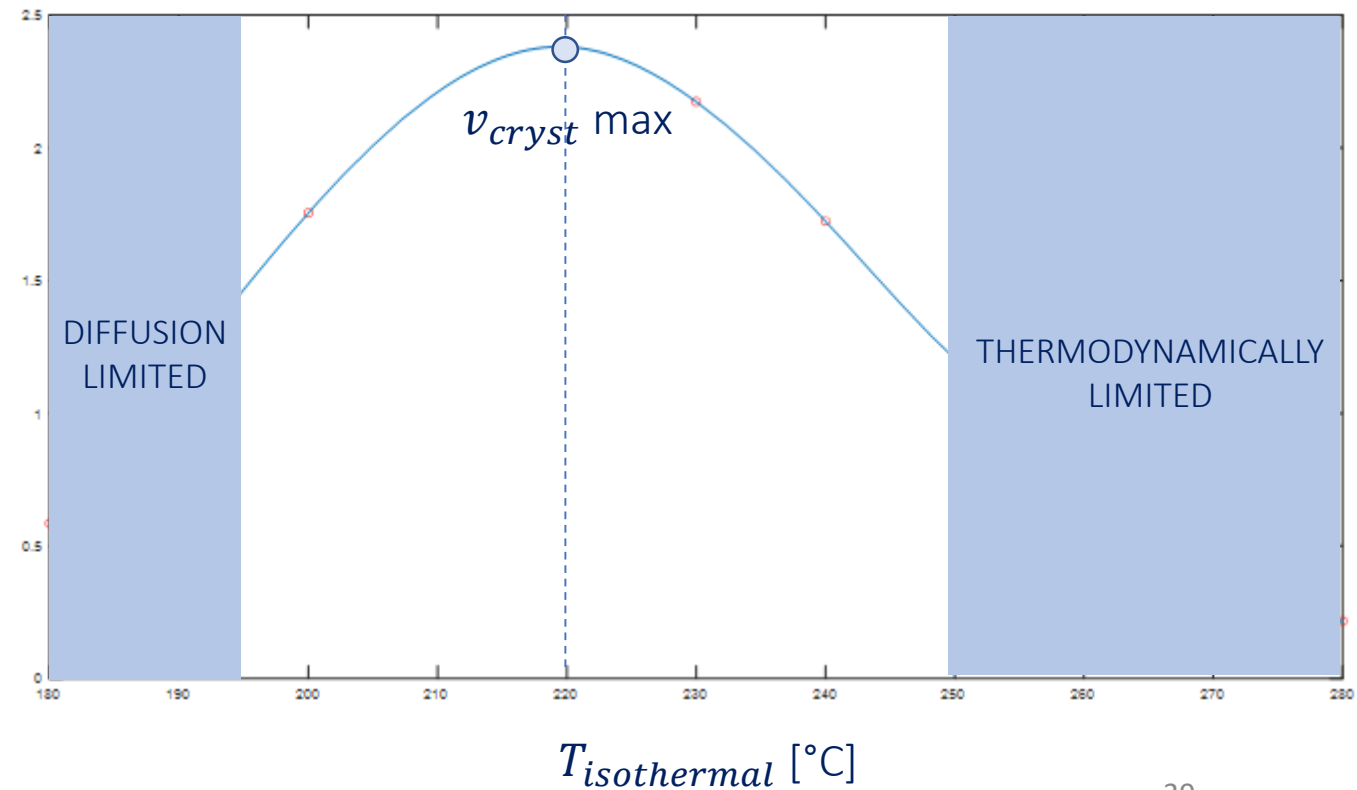
DEGREE OF CRYSTALLINITY – IMPACT OF COOLING RATE



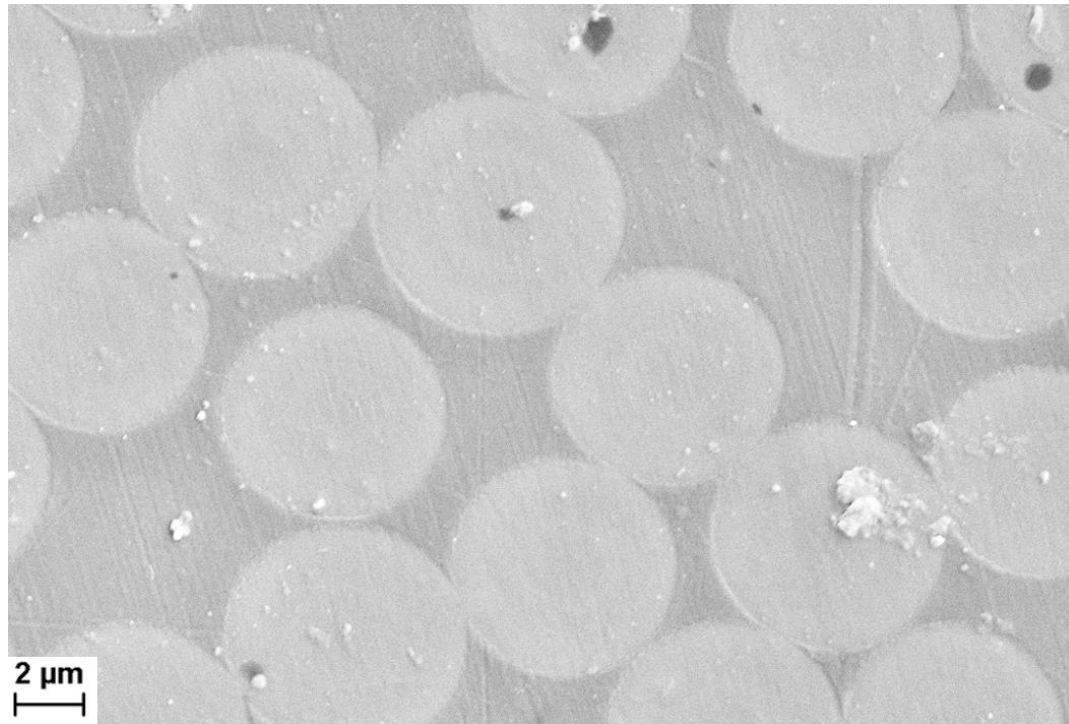
CRYSTALLISATION RATE – ISOTHERMAL CRYSTALLIZATION



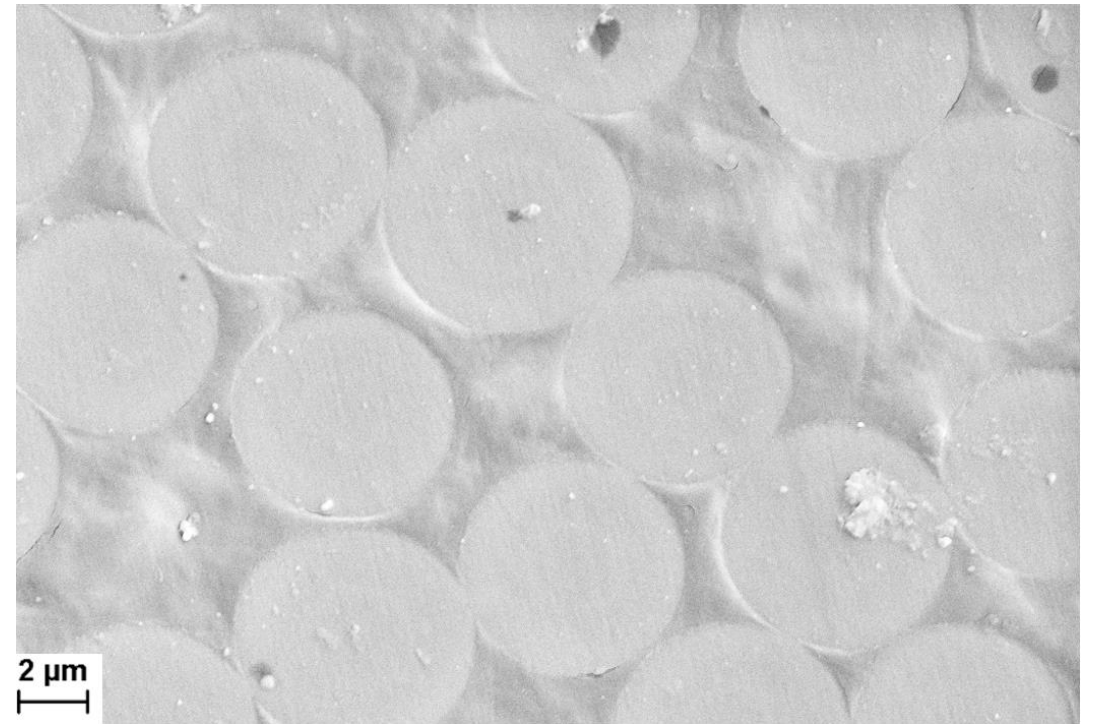
$$\frac{1}{t} [s^{-1}]$$



IMPACT OF ELECTRON BEAM ON THE MATRIX - SEM

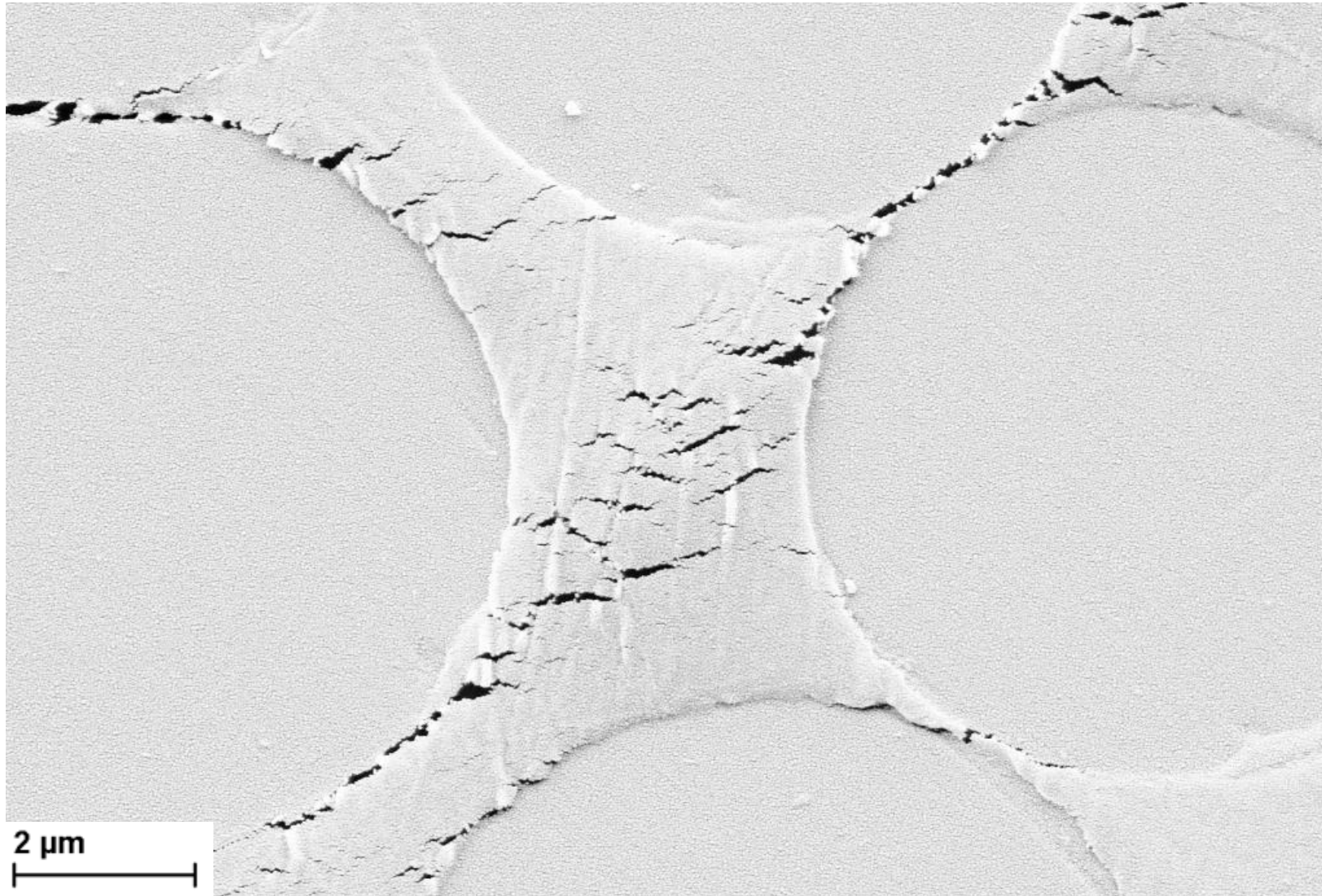


e- BEAM
→

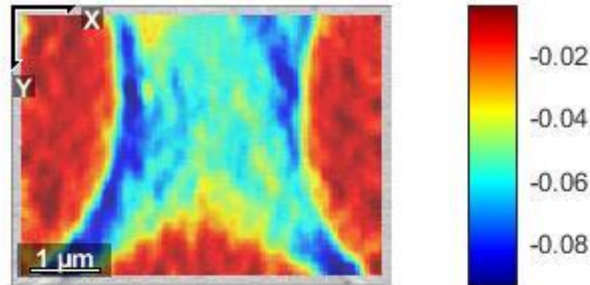


SEM + DIC (TRANSVERSE COMPRESSION TEST)

4 kV, SE2, Pt coating



SEM + DIC (TRANSVERSE COMPRESSION TEST) N_{corr}



Type: exx-plot

Reference Name: Sample01StrainAnalysis090322_01.jpg

Current Name: Sample01StrainAnalysis090322_02.jpg

Analysis type: regular

RG-DIC Radius: 12 | Strain Radius: 5 | Subset Spacing: 5

Diffnorm Cutoff: 1e-06 | Iteration Cutoff: 50 | Threads: 4

Step Analysis: Disabled

RG-DIC Subset Truncation: Enabled | Strain Subset Truncati

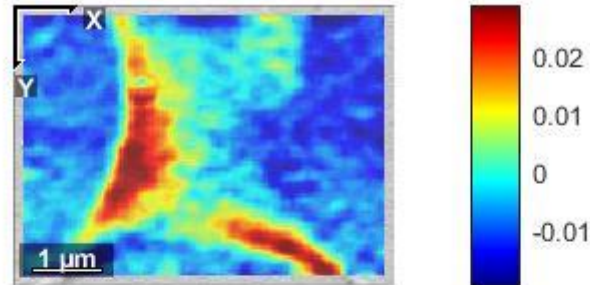
Image Correspondences: [0 1]

Units/pixels: 0.0059069 $\mu\text{m}/\text{pixels}$

Correlation Coefficient Cutoff: 1.5168

Radial Lens Distortion Coefficient: 0

Max: 0.0050 | Median: -0.0410 | Min: -0.1144



Type: exy-plot

Reference Name: Sample01StrainAnalysis090322_01.jpg

Current Name: Sample01StrainAnalysis090322_02.jpg

Analysis type: regular

RG-DIC Radius: 12 | Strain Radius: 5 | Subset Spacing: 5

Diffnorm Cutoff: 1e-06 | Iteration Cutoff: 50 | Threads: 4

Step Analysis: Disabled

RG-DIC Subset Truncation: Enabled | Strain Subset Truncati

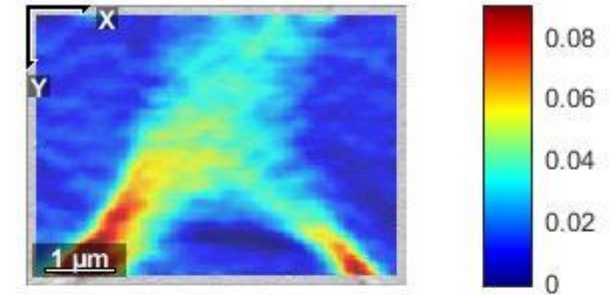
Image Correspondences: [0 1]

Units/pixels: 0.0059069 $\mu\text{m}/\text{pixels}$

Correlation Coefficient Cutoff: 1.5168

Radial Lens Distortion Coefficient: 0

Max: 0.0355 | Median: -0.0054 | Min: -0.0349



Type: eyy-plot

Reference Name: Sample01StrainAnalysis090322_01.jpg

Current Name: Sample01StrainAnalysis090322_02.jpg

Analysis type: regular

RG-DIC Radius: 12 | Strain Radius: 5 | Subset Spacing: 5

Diffnorm Cutoff: 1e-06 | Iteration Cutoff: 50 | Threads: 4

Step Analysis: Disabled

RG-DIC Subset Truncation: Enabled | Strain Subset Truncati

Image Correspondences: [0 1]

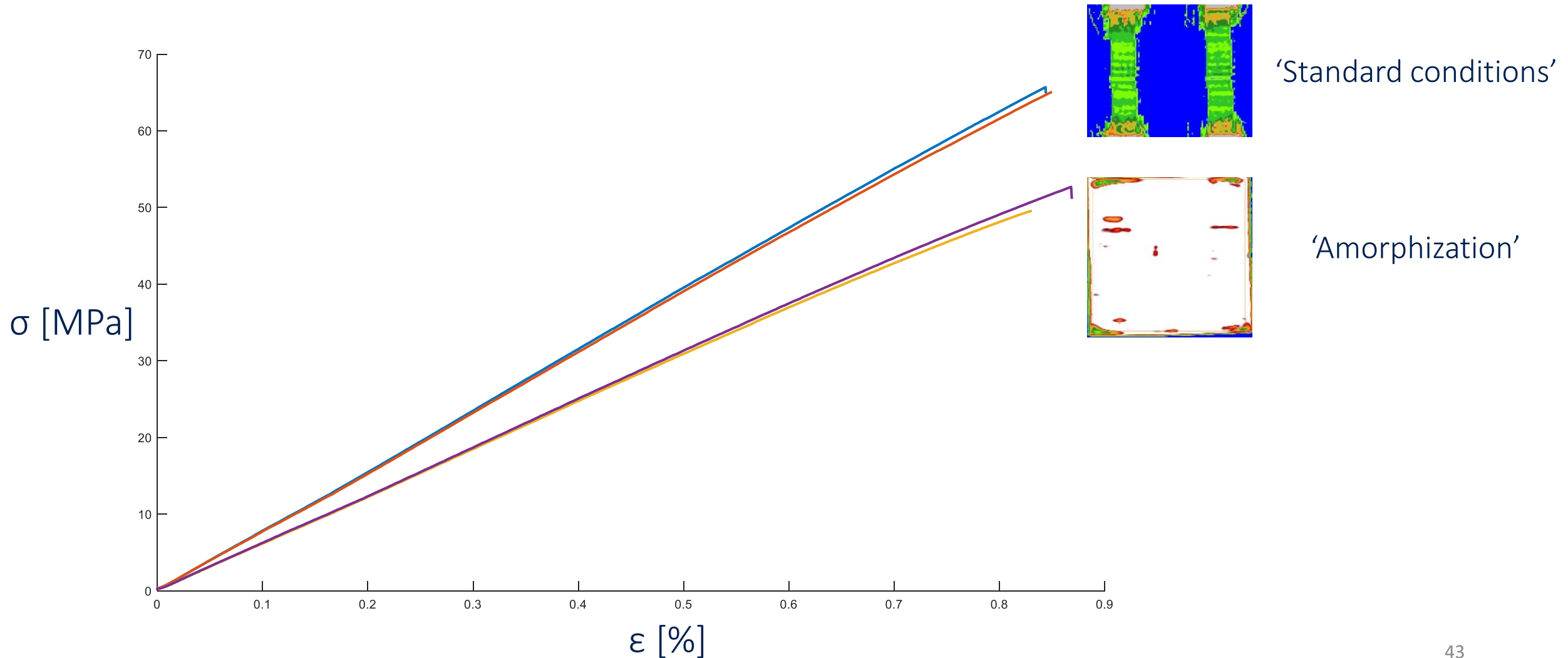
Units/pixels: 0.0059069 $\mu\text{m}/\text{pixels}$

Correlation Coefficient Cutoff: 1.5168

Radial Lens Distortion Coefficient: 0

Max: 0.1171 | Median: 0.0181 | Min: -0.0121

IMPACT OF PROCESSING CONDITIONS - TRANSVERSE TENSILE TESTS



IMPACT OF PROCESSING CONDITIONS - TRANSVERSE COMPRESSION TESTS

